Journal of Non-Crystalline Solids 114 (1989) 609–611 North-Holland

## OPTICALLY AND THERMALLY INDUCED REVERSIBLE CHANGES OF MIDGAP STATES IN UNDOPED a-Si:H

### Hideharu MATSUURA

Electrotechnical Laboratory, 1-1-4 Umezono, Tsukuba, Ibaraki 305, Japan

Midgap-state profiles in undoped hydrogenated amorphous silicon (a-Si:H) have been measured after light soaking, rapid cooling, and annealing using the heterojunction-monitored capacitance (HMC) method. After a short-time ( $\leq 4$  h) light soaking under AM1 with 100 mW/cm2 at room temperature, the increase in midgap states which were assigned to be singly occupied dangling bonds (D<sup>O</sup>) in asdeposited films was observed. The attempt-to-escape frequency ( $v_n$ ) for electrons of those states was about 7x10<sup>11</sup> s<sup>-1</sup>. Successive long-time ( $\leq 75$  h) light soaking, however, produced another midgap states with  $v_n$  of 2x10<sup>13</sup> s<sup>-1</sup>. Both light-induced metastable states with two kinds of  $v_n$  were located at around 0.85 eV below the conduction band edge. After annealing those samples up to 200 °C for 2 h, both states were completely recovered, but the recovering behavior was quite different from one with a small  $v_n$  to the other with a large  $v_n$ . On the other hand, the states produced by rapid cooling from 300 °C were similar to those induced by the short-time light soaking in their  $v_n$  as well as recovering behavior by the annealing.

# 1. INTRODUCTION

The existence of the light-induced reversible changes in hydrogenated amorphous silicon (a-Si:H) has attracted considerable attention, mostly because of the degradation of the efficiency of amorphous solar cells. This effect, which is known as the Staebler-Wronski effect,<sup>1</sup> arises from the creation of metastable defects by light soaking, and it is recovered by the annealing at 150-200 °C. Han and Fritzsche,<sup>2</sup> and Qiu et al.<sup>3</sup> reported that two kinds of light-induced states were produced by the light soaking; one was detected by photoconductivity (Δoph) measurements, and the other was observed by constant photocurrent measurements (CPM). Kumeda et al.<sup>4</sup> pointed out from the results of electron-spin-resonance (ESR) that singly occupied dangling bonds (D<sup>Q</sup>) created by the short-time (3 h) light soaking were easily annealed out, while D<sup>0</sup> created by the long-time (53 h) light soaking was resistant to the annealing. In this paper, changes of midgap-state profiles are investigated before and after the light soaking, the rapid cooling, and the annealing by means of the heterojunction-monitoredcapacitance (HMC) method<sup>5</sup> which is applicable to the undoped a-Si:H.

# 2. EXPERIMENT

Undoped a-Si:H / p-type crystalline silicon (p c-Si) heterojunctions were fabricated as follows. Undoped a-Si:H films (1.2-1.5  $\mu$ m thickness) were deposited by rf glow-discharge decomposition of pure SiH4 gas onto p c-Si substrates heated to T<sub>S</sub> = 200-300 °C. After turning off the plasma, the substrate temperature was kept as it was for 10 min. Then the specimen was cooled down slowly. The acceptor density (N<sub>A</sub>) of the p c-Si was 1x10<sup>16</sup> cm<sup>-3</sup>. Since magnesium (Mg) is known to form an ohmic contact with undoped a-Si:H,<sup>6</sup> Mg was evaporated on an area (0.785 mm<sup>2</sup>) of as-deposited films at room temperature (asdeposited films). For other heterojunctions, Mg was evaporated at room temperature after a-Si:H films were exposed to AM1 light with 100 mW/cm<sup>2</sup> at room temperature (light-soaked films), or after those films, which were kept at a given high temperature (T<sub>RC</sub>) in H<sub>2</sub> atmosphere for 10 min, were immediately dropped into liquid nitrogen (rapidly-cooled films).

The midgap-state density (N<sub>I</sub>) of undoped a-Si:H was estimated from the high-frequency (1 MHz) capacitance-voltage characteristics at room temperature using the steady-states HMC method.<sup>5</sup> The HMC signal H(t), which is obtained from transient capacitance of the heterojunction measured at 2 MHz, approximately corresponds to the density-of-states distribution [g(E)] through g(E)=H(t)/kT, and the energy location below the conduction band edge (E<sub>C</sub>) is expressed by E<sub>C</sub>-E=kT/n (v<sub>n</sub>t), where k is the Boltzmann's constant, T is the absolute measuring temperature, t is the time after the reverse bias is applied to the junction, v<sub>n</sub> is the attempt-to-escape frequency for electrons which can be estimated from the temperature dependence of the time (t<sub>p</sub>) at the peak of H(t).<sup>5</sup>

#### 3. RESULTS AND DISCUSSION

#### 3.1. Light soaking

The value of N<sub>I</sub> increased with between one-second and one-third powers of the illumination time (t<sub>I</sub>L), whose behavior was quite similar to the results obtained from the ESR measurement. The value of the activation energy ( $\Delta E_{ac}$ ) of dark conductivity of the film was independent of t<sub>I</sub>L when t<sub>I</sub>L was longer than 3 h, although  $\Delta E_{ac}$  of light-soaked films was larger than that of asdeposited films. The H(t) signals did not change after the asdeposited film was annealed at 200 °C for 2 h in a vacuum, and the H(t) signals for as-deposited films were quite similar to those for the films annealed at 200 °C for 2 h after the light soaking. After the short-time (≤ 4 h) light soaking, the magnitude of H(t) increased without any shift of t<sub>p</sub>, while after the long-time (≥ 75 h) light soaking t<sub>p</sub> was shifted toward shorter time and kept

constant, as is clearly shown in Fig. 1. From the temperature dependence of  $t_p$ ,  $v_n$  and the energy position  $(E_p)$  corresponding to the peak of H(t) were estimated as follows;  $v_{n1}$  and  $(E_C-E_{p1})$  for the as-deposited and the short-time light-soaked films were about  $7x10^{11}$  s<sup>-1</sup> and 0.86 eV, respectively, and  $v_{n2}$  and  $(E_C-E_{p2})$  for the long-time light-soaked films were about  $2x10^{13}$  s<sup>-1</sup> and 0.84 eV, respectively. Although the change of  $t_p$  can, as one possibility, be thought to arise from the change of the free electron concentration in the depletion region of a-Si:H which leads to the change in the reverse current of the heterojunction,<sup>7</sup> the reverse current for the long-time light-soaked film was the same as that for the as-deposited film, indicating that the change of  $t_p$  must be originated from the change of  $v_p$ . Figure 2 schematically summarized the above results.

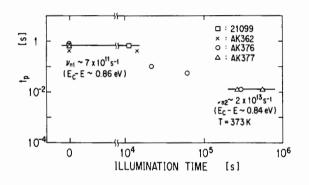
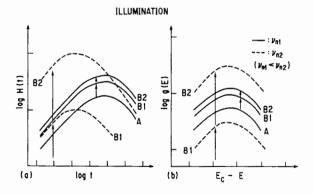


FIGURE 1 The dependence of the time  $(t_p)$  at the peak of H(t) on illumination time. The solid lines are guides to eye.



#### FIGURE 2

Schematic changes of midgap states by light soaking. The solid and dashed lines represent the states with small and large  $v_n$ respectively. A, B1, and B2 correspond to the as-deposited (completely annealed), the short-time light-soaked, and the long-time light-soaked films, respectively. (b) was estimated from (a) using the relations of g(E)=H(t)/kT and E\_C=kT *in* ( $v_n$ t).

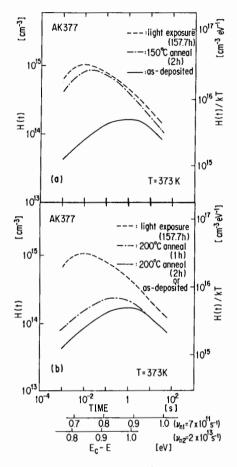
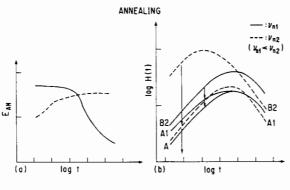


FIGURE 3 Changes of H(t) by annealing for the long-time light-soaked film.

### 3.2. Annealing of light-soaked films

Figure 3(a) shows the changes of H(t) for the long-time lightsoaked films by a 150 °C annealing for 2 h, and Fig. 3(b) shows those by a 200 °C annealing. Since the H(t) signals include information on two kinds of midgap states, two sorts of energy scales corresponding to vn1 and vn2 are shown in the abscissa below the time scale. We previously studied the annealing kinetics using the short-time light-soaked films,<sup>8</sup> from which monomolecular annealing kinetics were found to be suitable for explaining the results. In Fig. 3(a), the change in H(t) at t>10 s must be originated from the change of the states having vn1 because the activation energy for annealing (EAN) estimated from the figure has the same behavior as EAN obtained from the short-time light-soaked film,<sup>8</sup> while the change in H(t) at t<10<sup>-1</sup> s is thought to arise from the change of the states having vn2 because those states are dominant in the region. Since the energy dependence of EAN for the short-time light-soaked films is known,<sup>8</sup> changes in EAN and H(t) for the annealing process in



**FIGURE 4** 

Activation energies for the annealing and schematic changes of midgap states by the annealing. The solid and dashed lines represent the states with small and large  $v_n$ , respectively. B2, A1, and A correspond to the long-time light-scaked, the short-time annealed, and the completely annealed films, respectively.

the long-time light-soaked films can be schematically described as shown in Fig. 4.

### 3.3. Rapid cooling

In the film deposited at 200 °C, N<sub>I</sub> in the rapidly-cooled film for  $T_{RC} = 200$  °C was equal to that (about  $10^{16}$  cm<sup>-3</sup>) in the asdeposited film, N<sub>I</sub> for  $T_{RC} = 250$  °C decreased as low as N<sub>I</sub> (about  $5\times10^{15}$  cm<sup>-3</sup>) in a good quality film, and then N<sub>I</sub> increased with an increase of  $T_{RC}$ . In the good quality films deposited at 250 and 300 °C, on the other hand, N<sub>I</sub> did not change in the range of  $T_{RC} \le 250$  °C, and then N<sub>I</sub> increased with a further increase of  $T_{RC}$ . From the transient HMC method, the states increased by rapid cooling from 300 °C were the states having vn1.

## 3.4. Correspondence with other results

Both states produced by the light soaking, which are distinguished by the difference in  $v_{n}$  must be originated from spin centers because the behavior of N<sub>I</sub> by the light soaking is quite similar to that obtained from ESR. From the study of asdeposited films,<sup>5</sup> the states with  $v_{n1}$  were found to be D<sup>0</sup>, and those states could also be thermally created because they exist in the as-deposited and the rapidly-cooled films. On the other hand, the other states with  $v_{n2}$  are noted as DL<sup>0</sup>, since the states can be created only by the light soaking.

Han and Fritzsche,<sup>2</sup> and Qiu et al.<sup>3</sup> reported that two kinds of metastable states could be produced by light soaking. The first light-induced reversible states had a small capture-cross section ( $\sigma_{n1}$ ) for electrons, and were detected by CPM. The second states had a large capture-cross section ( $\sigma_{n2}$ ) for electrons, and were detected by  $\Delta \sigma_{ph}$  using a small incident light flux with a 2-eV photon energy. The value of  $\sigma_{n1}$  was smaller by about one order of magnitude than  $\sigma_{n2}$ , and both states were located in

the midgap. The value of  $\Delta \sigma_{ph}$  decreased by the short-time (3-4 h) light soaking at 100 K, while an absorption coefficient ( $\alpha_{sub}$ ) at 1.0 eV obtained by CPM did not change. By the same light soaking at 300 K, however,  $\Delta \sigma_{ph}$  decreased and  $\alpha_{sub}$ increased. These indicate that the density of the first states (N1) increases only at high temperature while the density of the second states (N<sub>2</sub>) increases at any temperature because  $\Delta \sigma_{ph}$ and a<sub>sub</sub> are expressed as ∆oph∝1/(on1N1+on2N2) and  $\alpha_{sub} \propto N_1 + N_2$ , respectively and  $\sigma_{n1} < \sigma_{n2}$ . In the light of our results, the first states correspond to D<sup>0</sup> just as they concluded, and the second states should correspond to  $DL^0$  because  $v_n$  is proportional to  $\sigma_n$ . The origin of DL<sup>0</sup> is still an open question, although Okushi et al.<sup>9</sup> have insisted the model in which the dangling-bond-like centers are produced by a spatially intimated coupling of pairs between dangling bonds and positively ionized impurities.

### 5. CONCLUSION

The midgap states having a small  $v_n$  were optically as well as thermally created, while the midgap states having a large  $v_n$  were created only by light soaking. Both states were located around 0.85 eV below the conduction band edge.

# ACKNOWLEDGEMENT

The author wishes to thank the members of the staff in the Electrotechnical Laboratory for their valuable insights and comments.

### REFERENCES

- D. L. Staebler and C. R. Wronski, Appl. Phys. Lett. 31 (1977) 292.
- D. Han and Fritzsche, J. Non-Cryst. Solids 59&60 (1983) 397.
- C. Qiu, W. Li, D. Han, and J. Pankove, J. Appl. Phys. 64 (1988) 713.
- M. Kumeda, H. Yokomichi, A. Morimoto, and T. Shimizu, Jpn. J. Appl. Phys. 25 (1986) L654.
- 5. H. Matsuura, J. Appl. Phys. 64 (1988) 1964.
- H. Matsuura, T. Okuno, H. Okushi, S. Yamasaki, A. Matsuda, N. Hata, H. Oheda, and K. Tanaka, Jpn. J. Appl. Phys. 22 (1983) L197.
- D. V. Lang, J. D. Cohen, J. P. Harbison, M. C. Chen, and A. M. Sergent, J. Non-Cryst. Solids 66 (1984) 217.
- 8. H. Matsuura, Appl. Phys. Lett. 54 (1989) 344.
- H. Okushi, R. Banerjee, and K. Tanaka, Advances in Disordered Semiconductors, Vol. 1, ed. H. Fritzsche (World Scientific, Singapore, 1989), pp. 657-685.