

Mechanisms of changes of hole  
concentration in Al-doped 6H-SiC by  
electron irradiation and annealing

Hideharu Matsuura, Hideki Yanagisawa, Kozo Nishino,  
Yoshiko Myojin, Takunori Nojiri, Yukei Matsuyama  
*Osaka Electro-Communication University*

Takeshi Ohshima  
*Japan Atomic Energy Agency*

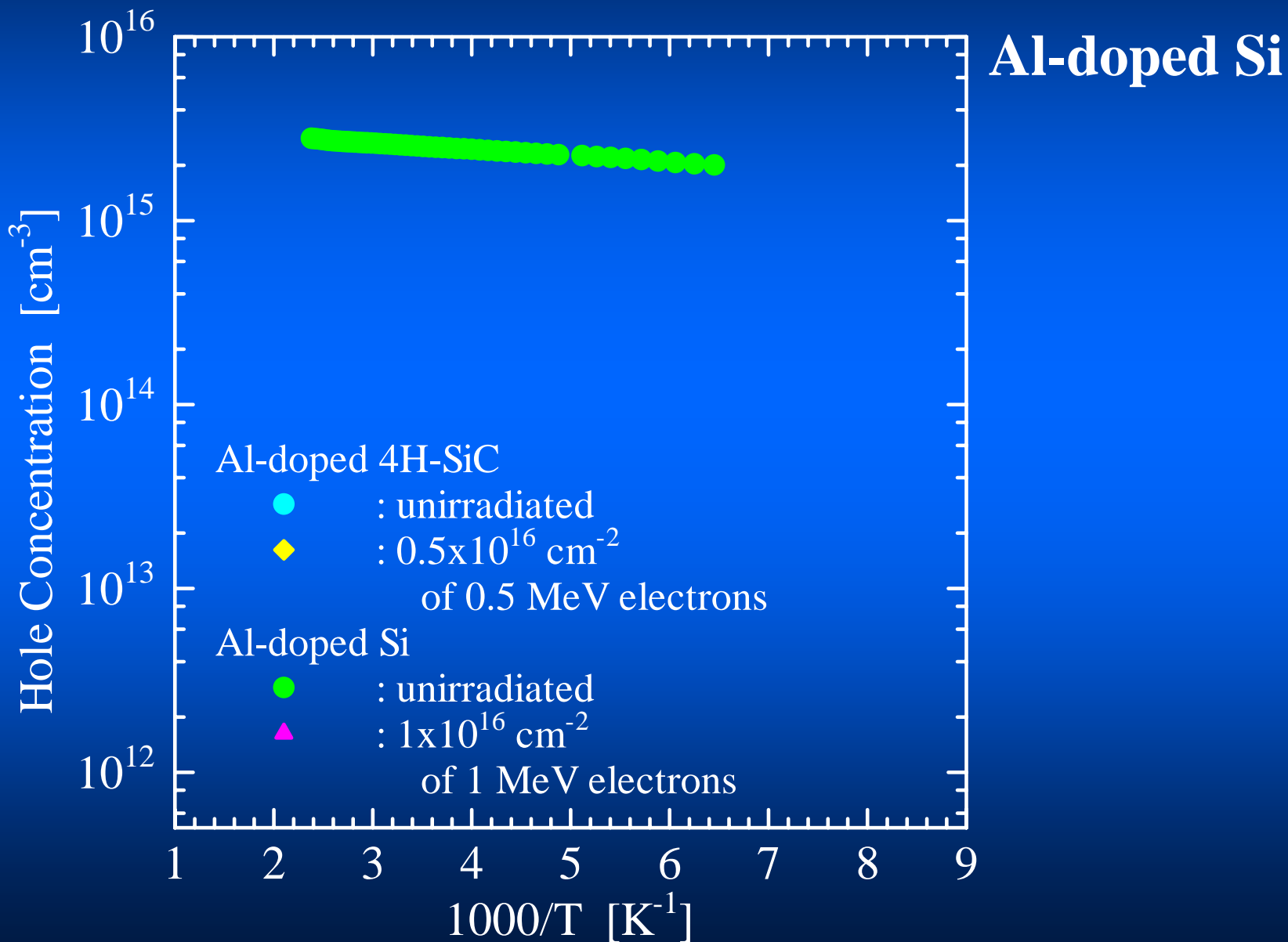
International Conference on Defects in Semiconductors(ICDS-25)  
20 July, 2009, St. Petersburg, Russia

# Background of our study

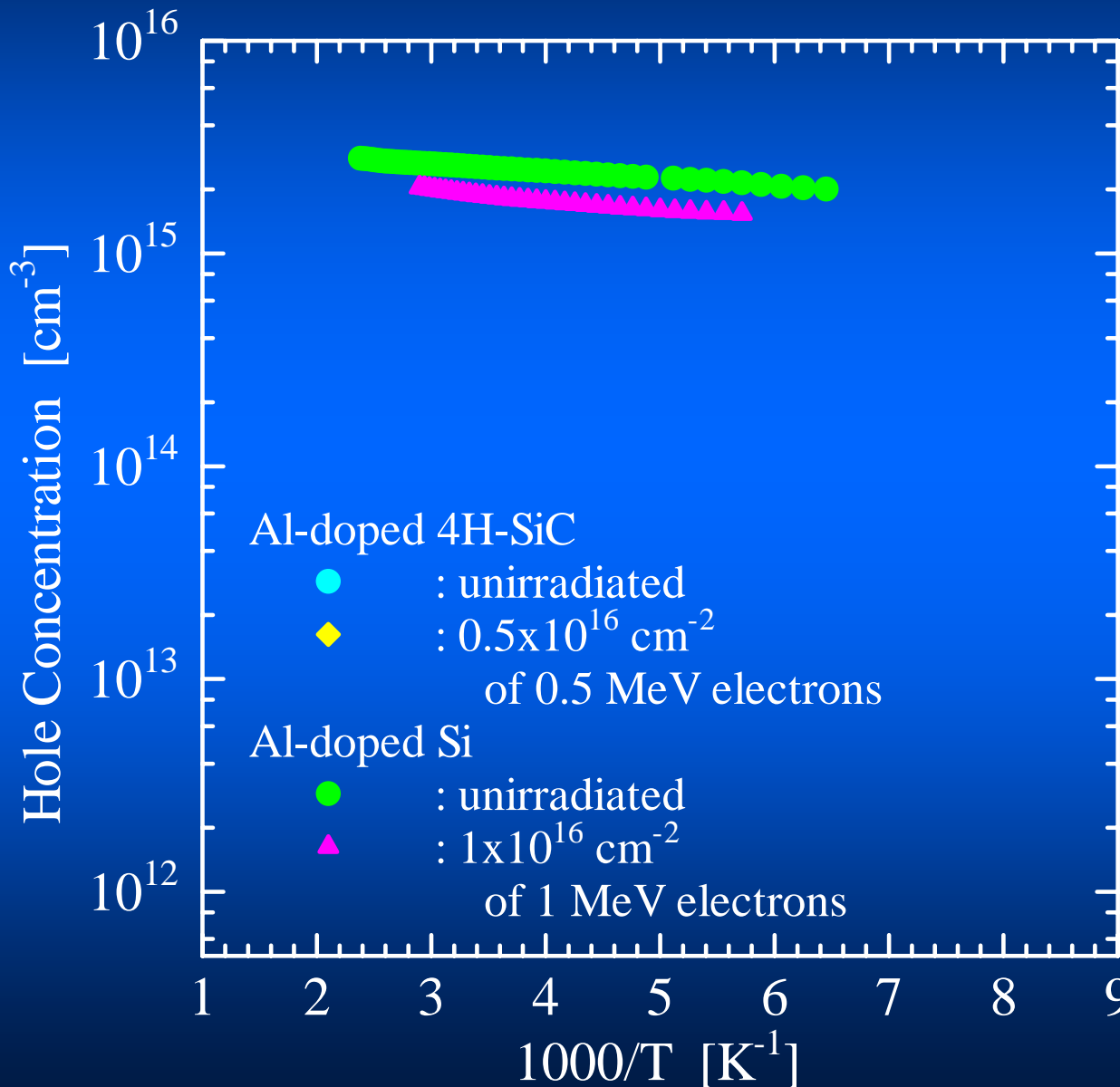
Silicon carbide (SiC) is a promising wide bandgap semiconductor for fabricating high-power and high-frequency electronic devices capable of operating at elevated temperature under radiation environment.

In order to understand the radiation-degradation of SiC electronic devices, changes of properties in SiC by radiation should be investigated.

# Comparison of radiation resistance of Si and SiC



# Comparison of radiation resistance of Si and SiC



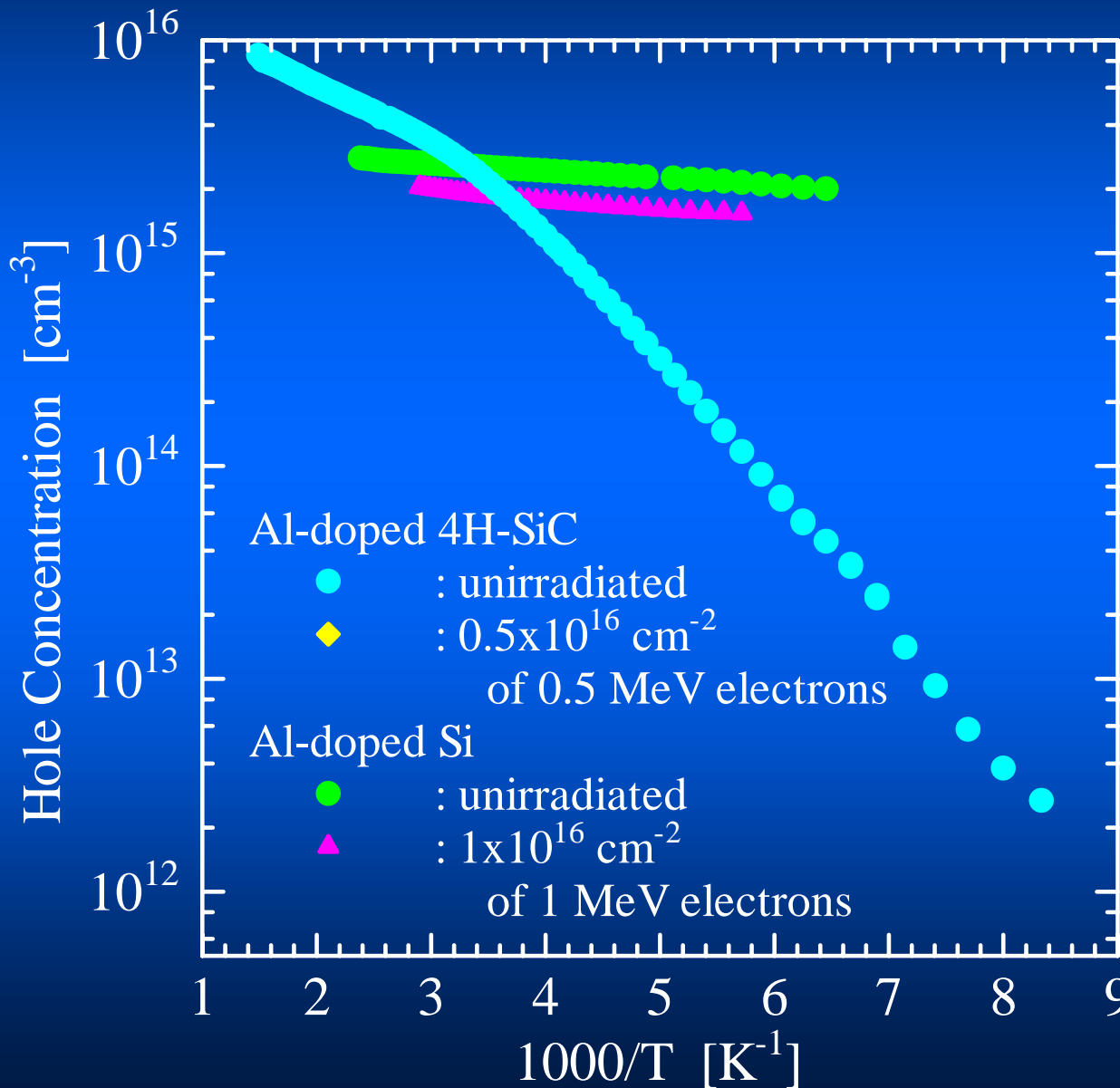
## Al-doped Si

Irradiation conditions  
1 MeV electrons  
 $1 \times 10^{16} \text{ cm}^{-2}$  fluence



p(T) decreased slightly.

# Comparison of radiation resistance of Si and SiC



## Al-doped Si

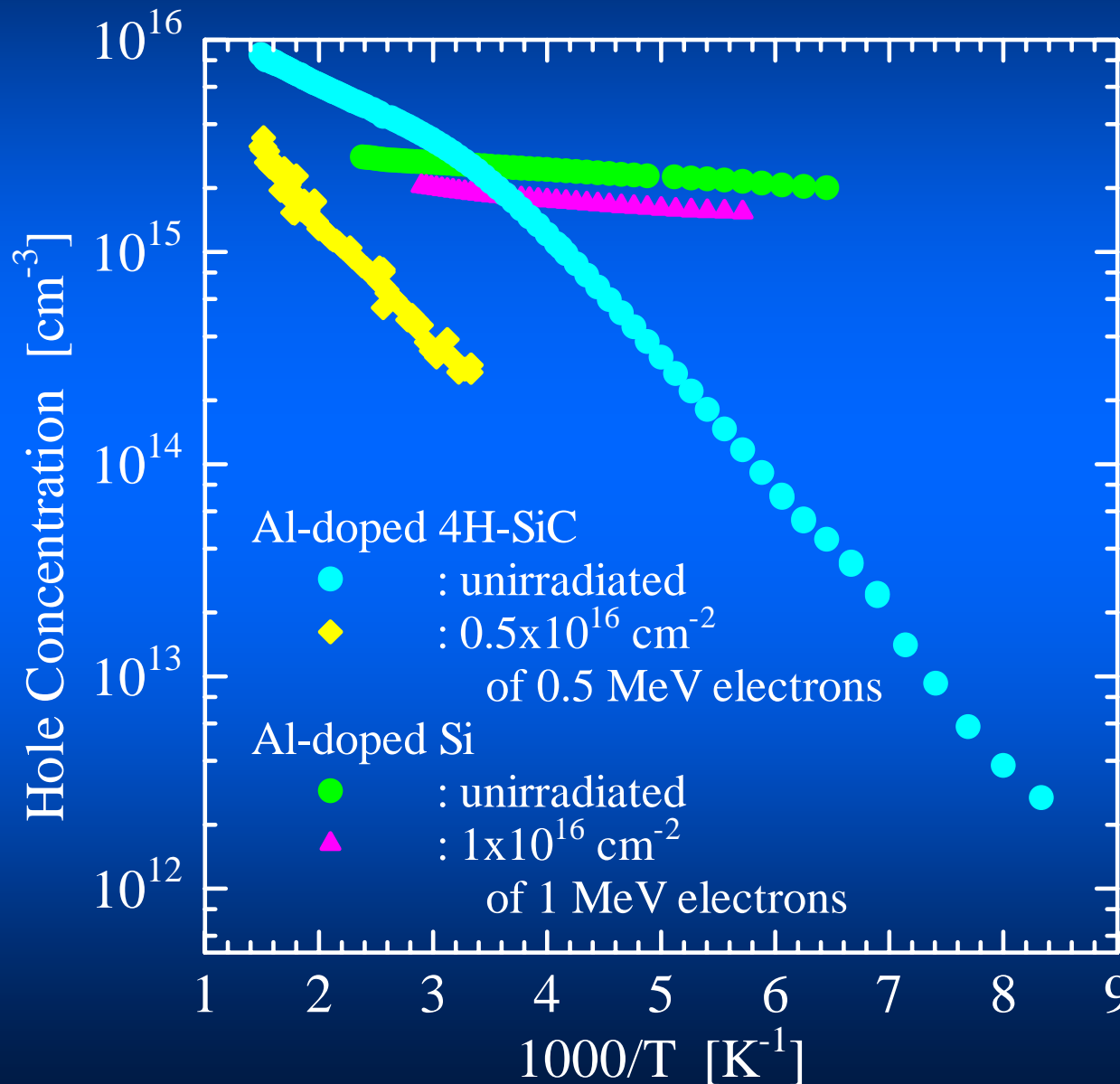
Irradiation conditions  
 $1 \text{ MeV}$  electrons  
 $1 \times 10^{16} \text{ cm}^{-2}$  fluence



p(T) decreased slightly.

## Al-doped 4H-SiC

# Comparison of radiation resistance of Si and SiC



Al-doped Si

Irradiation conditions  
1 MeV electrons  
 $1 \times 10^{16} \text{ cm}^{-2}$



p(T) decreased slightly.

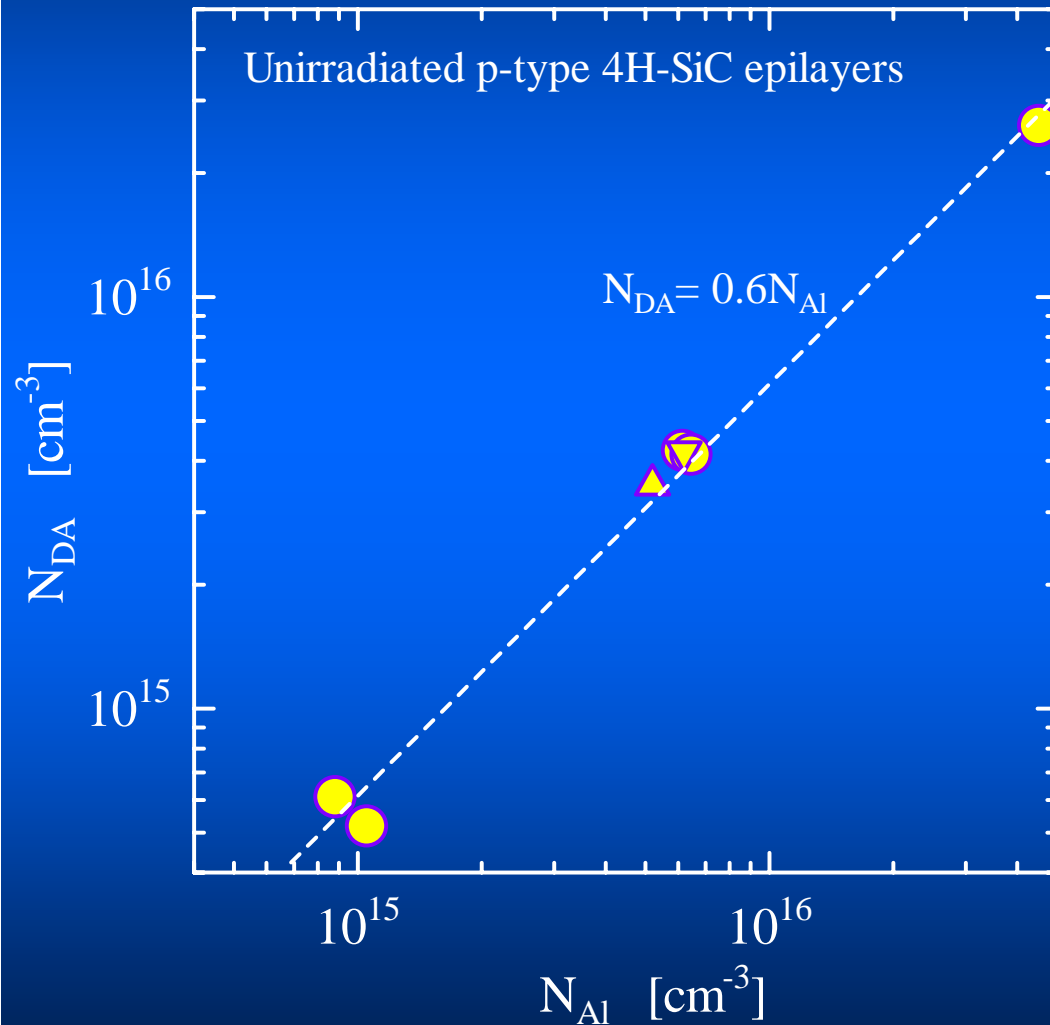
Al-doped 4H-SiC

Irradiation conditions  
0.5 MeV electrons  
 $0.5 \times 10^{16} \text{ cm}^{-2}$  fluence

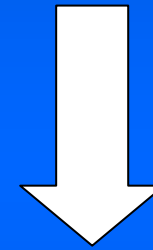


p(T) decreased  
**significantly.**

# Relationship between $N_{Al}$ and $N_{DA}$ in unirradiated Al-doped 4H-SiC epilayer

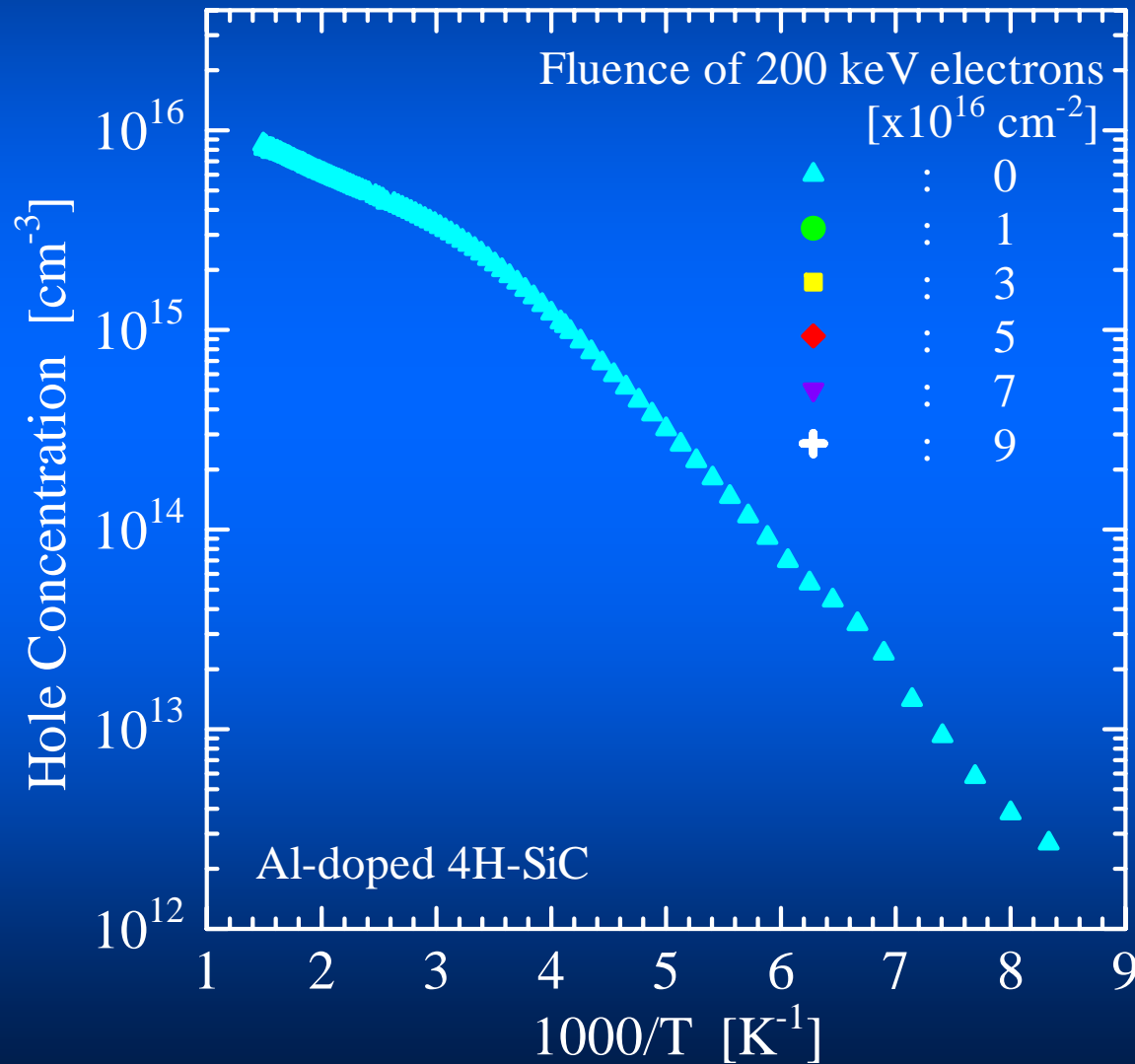


$$N_{DA} = 0.6 \times N_{Al}$$



**The unknown deep acceptor is most likely related to Al**

Reduction in  $p(T)$  in Al-doped p-type 4H-SiC  
by **200 keV** electron irradiation

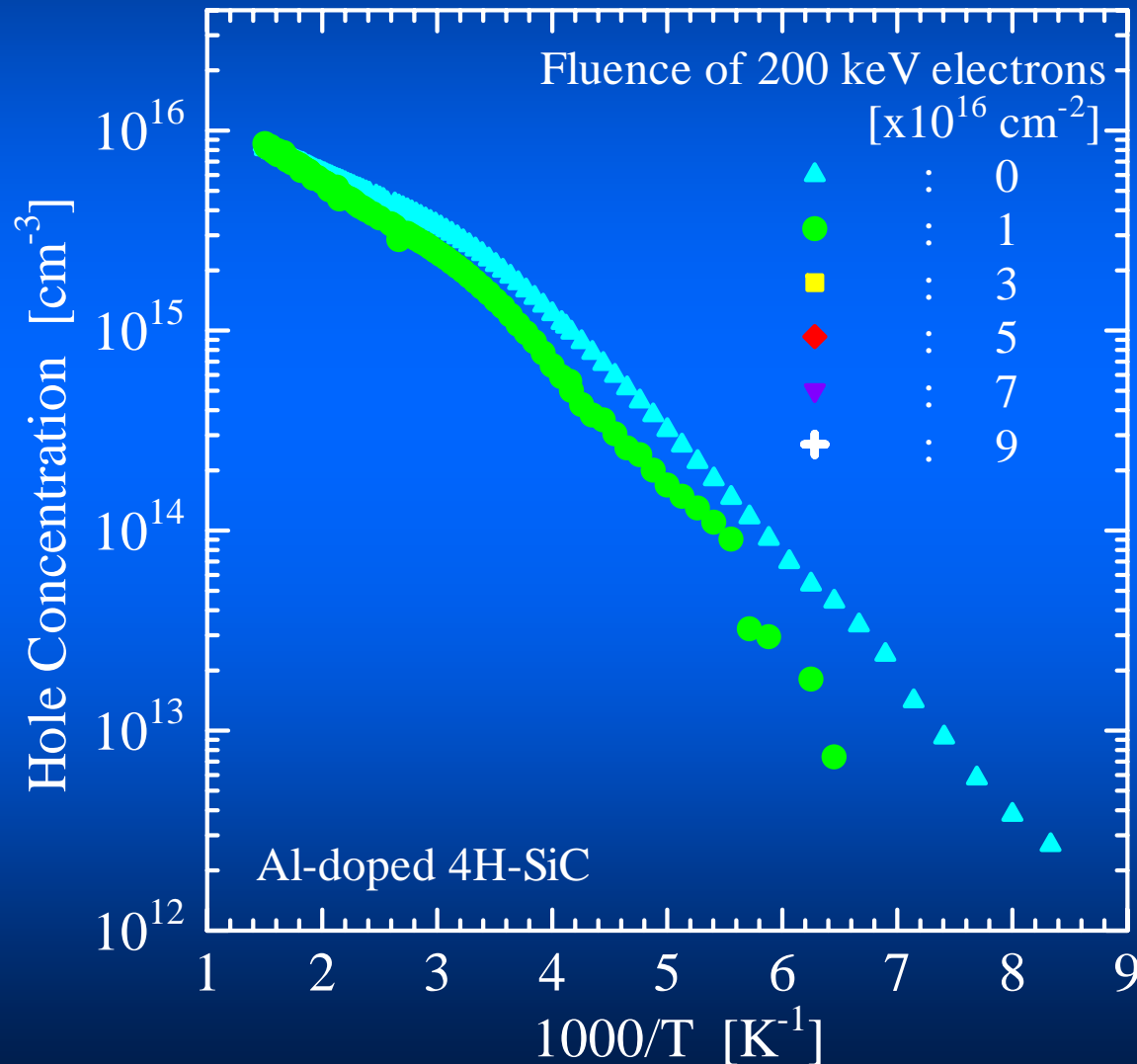


Fluence [cm<sup>-2</sup>]

0



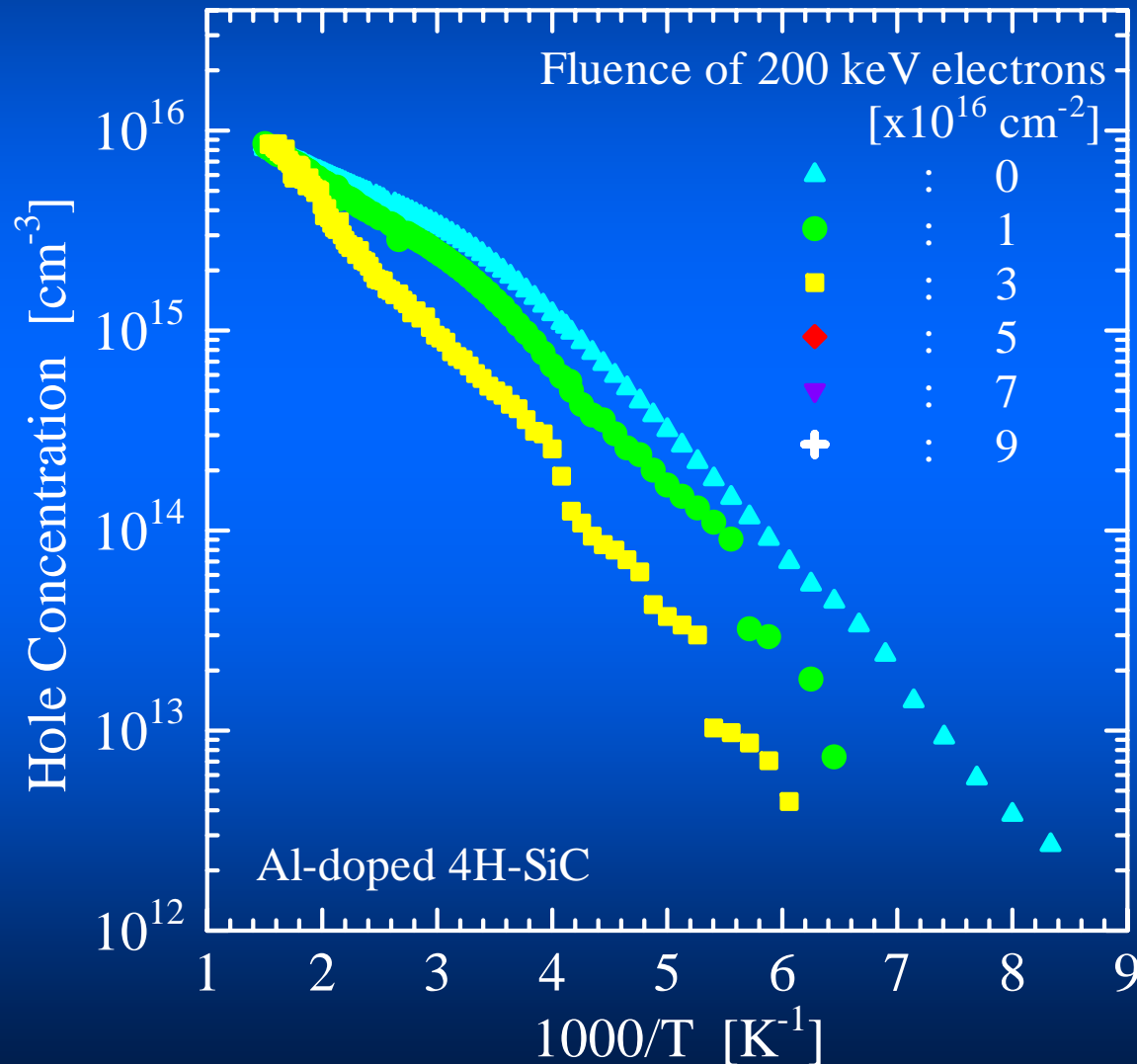
Reduction in  $p(T)$  in Al-doped p-type 4H-SiC  
by **200 keV** electron irradiation



Fluence [ $\text{cm}^{-2}$ ]

$1 \times 10^{16}$

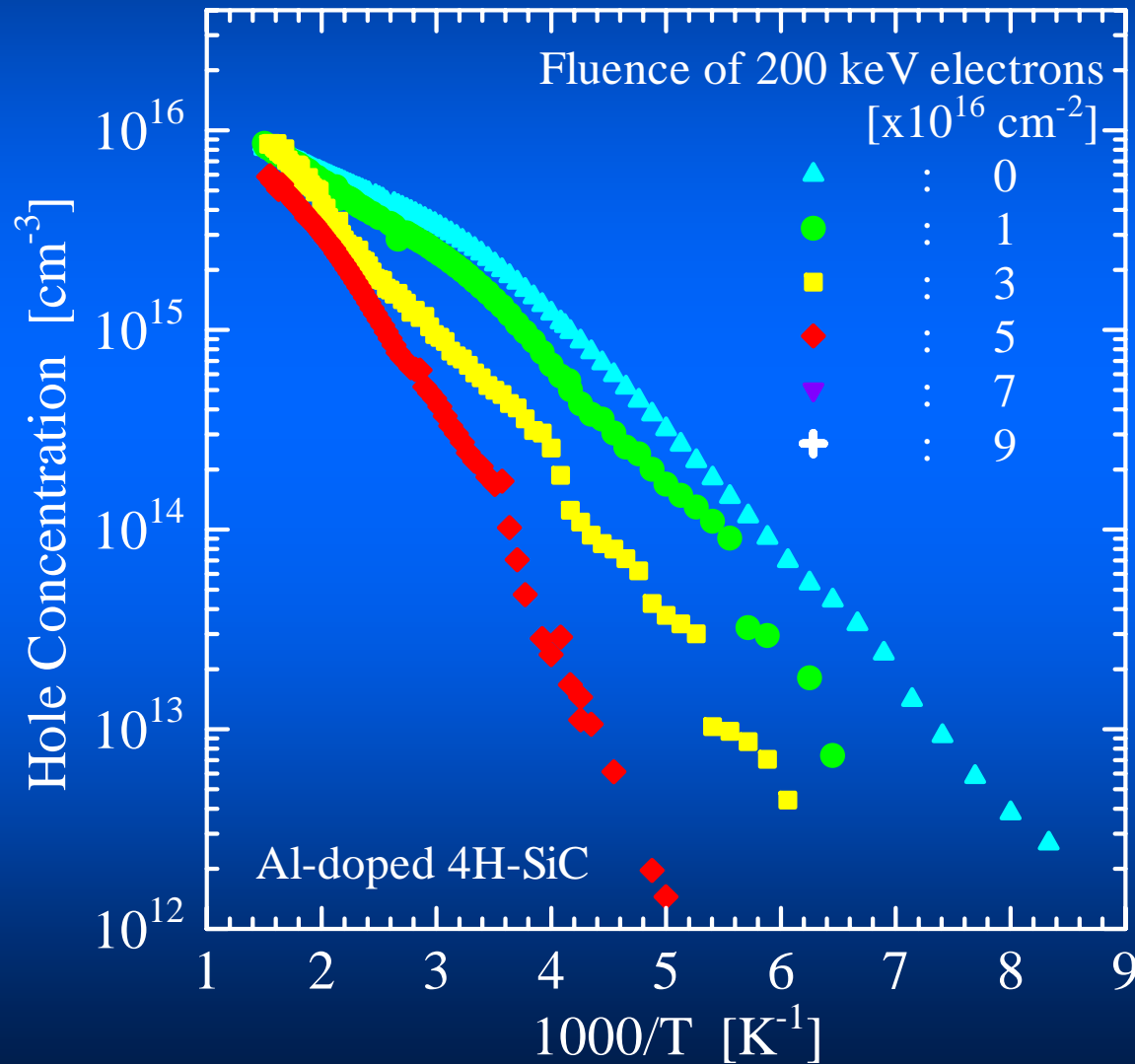
Reduction in  $p(T)$  in Al-doped p-type 4H-SiC  
by **200 keV** electron irradiation



Fluence [ $\text{cm}^{-2}$ ]

$3 \times 10^{16}$

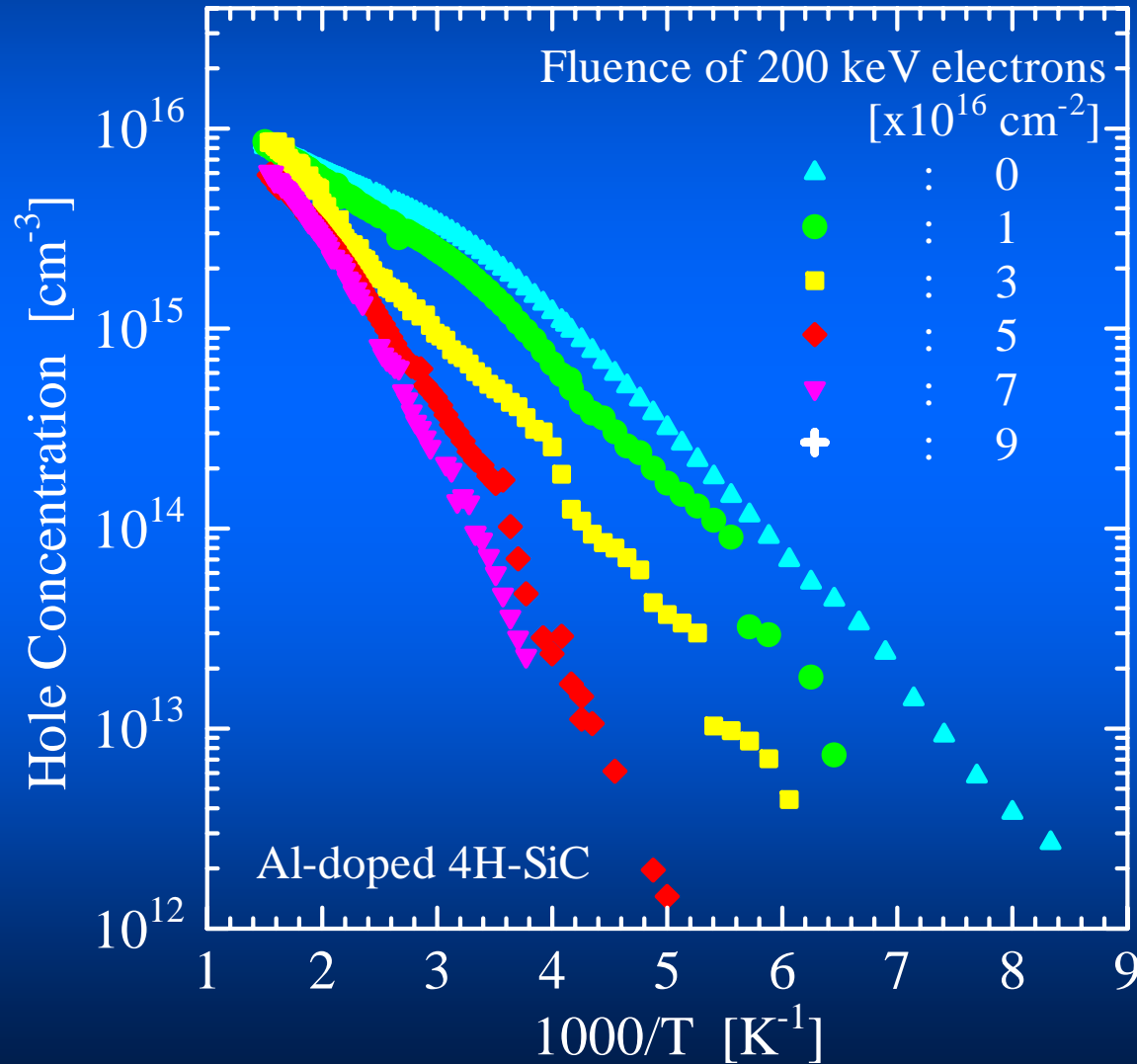
Reduction in  $p(T)$  in Al-doped p-type 4H-SiC  
by **200 keV** electron irradiation



Fluence [ $\text{cm}^{-2}$ ]

$5 \times 10^{16}$

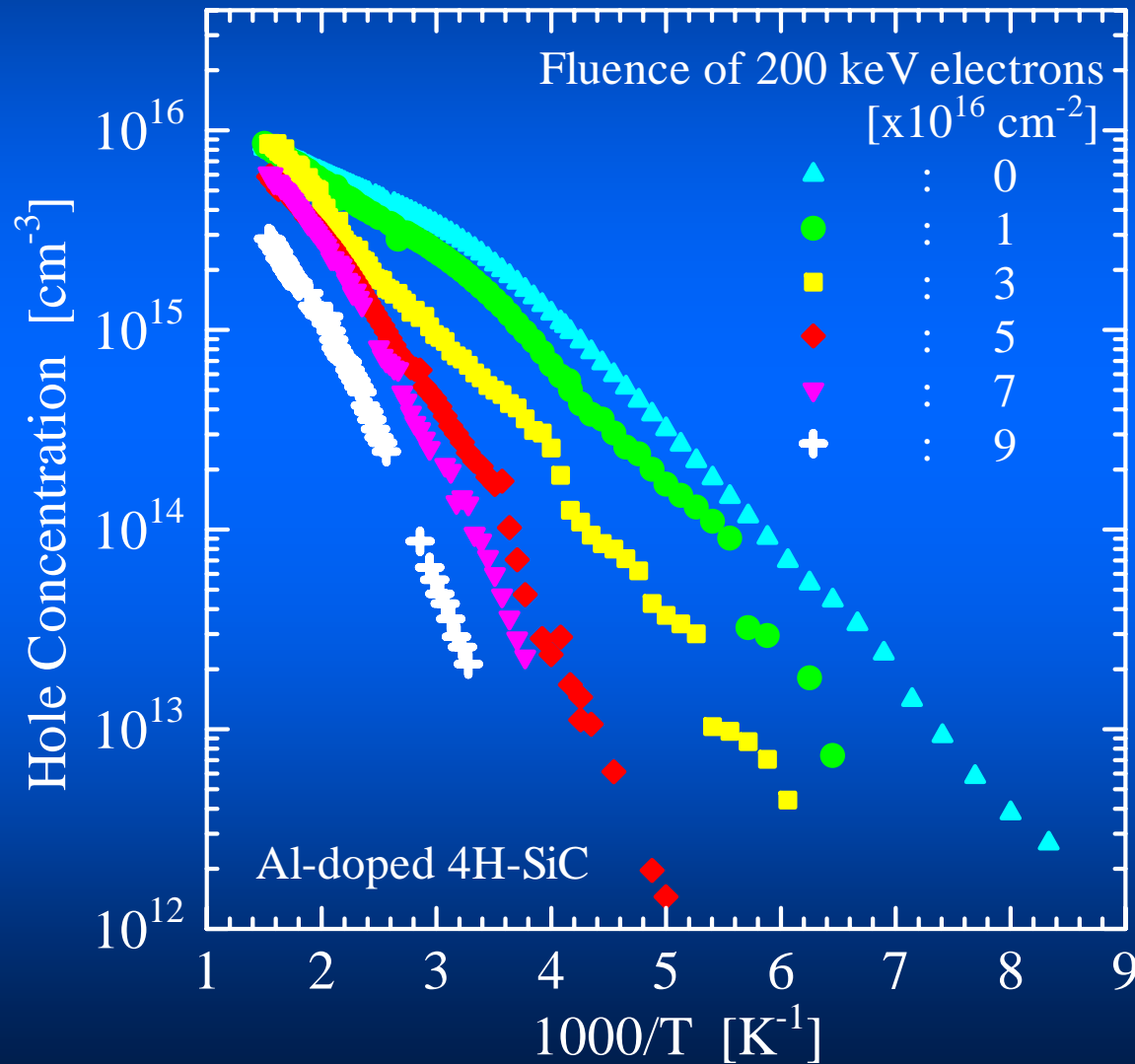
Reduction in  $p(T)$  in Al-doped p-type 4H-SiC  
by **200 keV** electron irradiation



Fluence [ $\text{cm}^{-2}$ ]

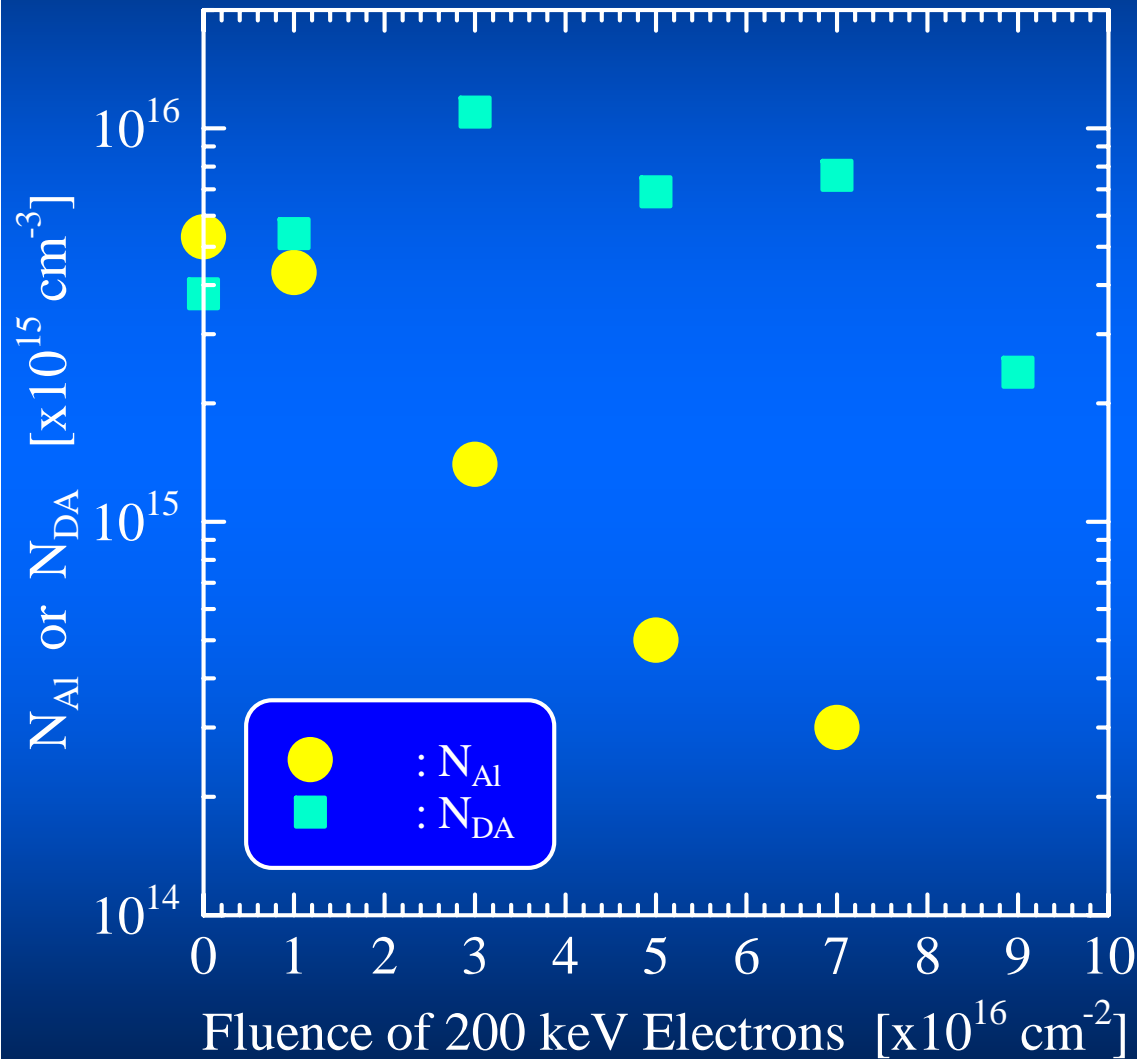
$7 \times 10^{16}$

Reduction in  $p(T)$  in Al-doped p-type 4H-SiC  
by **200 keV** electron irradiation

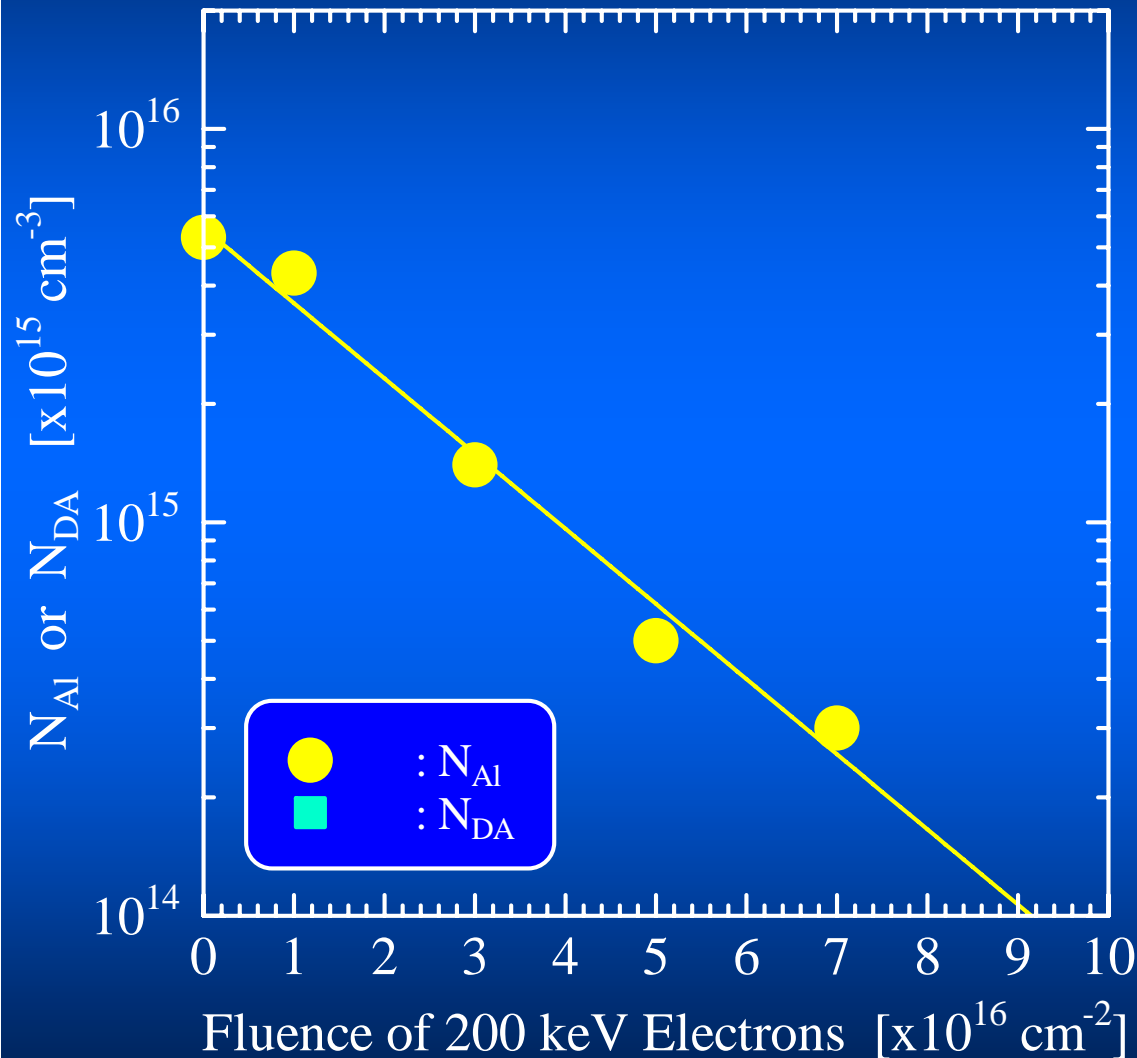


Fluence [ $\text{cm}^{-2}$ ]  
 $9 \times 10^{16}$

# Fluence Dependence of $N_{\text{Al}}$ and $N_{\text{DA}}$ in 4H-SiC



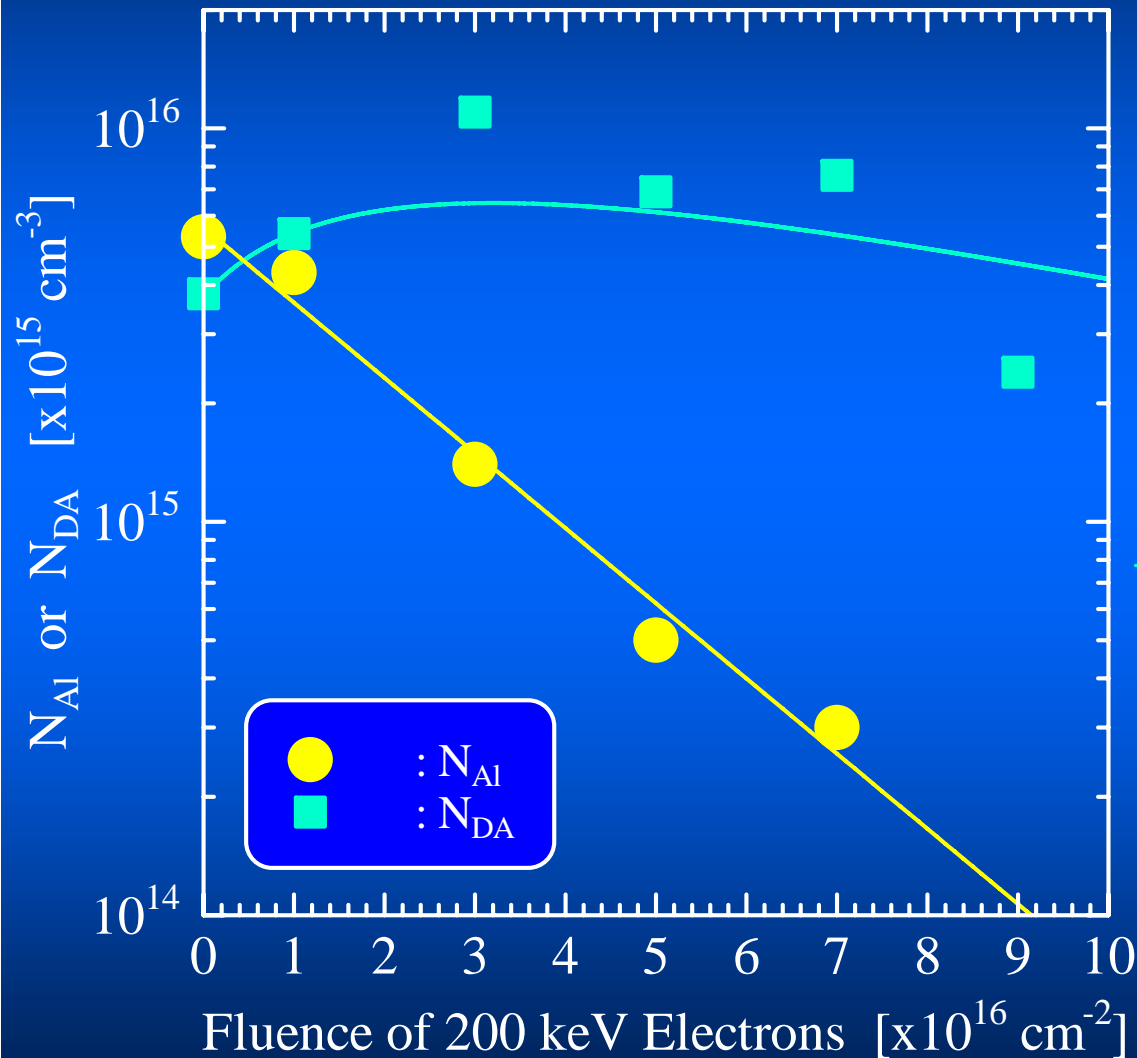
# Fluence Dependence of $N_{Al}$ and $N_{DA}$ in 4H-SiC



$$\frac{dN_{Al}(\Phi)}{d\Phi} = -\kappa_{Al} N_{Al}(\Phi)$$

$$\kappa_{Al} = 4.4 \times 10^{-17} \text{ cm}^2$$

# Fluence Dependence of $N_{Al}$ and $N_{DA}$ in 4H-SiC



$$\frac{dN_{Al}(\Phi)}{d\Phi} = -\kappa_{Al} N_{Al}(\Phi)$$

$$\kappa_{Al} = 4.4 \times 10^{-17} \text{ cm}^2$$

$$\frac{dN_{DA}(\Phi)}{d\Phi} = -\frac{dN_{Al}(\Phi)}{d\Phi} - \kappa_{DA} N_{DA}(\Phi)$$

$$\kappa_{DA} = 1.0 \times 10^{-17} \text{ cm}^2$$



## Motivation of our study

1. Which polytype of SiC is radiation-resistant?

**How about Al-doped 6H-SiC**

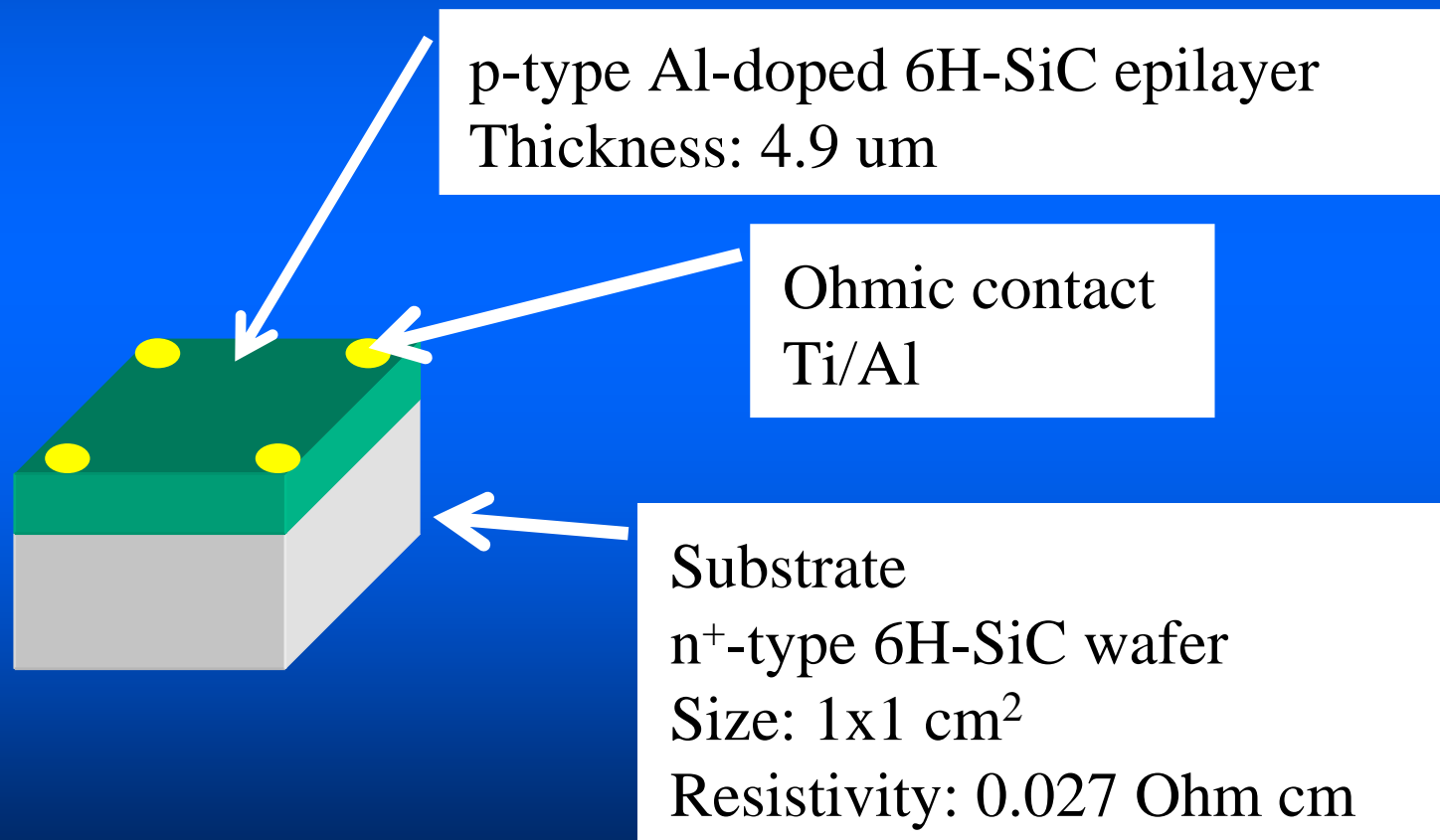
2. Why is the  $p(T)$  reduced by irradiation?

**Increase in compensating density or  
Change of acceptor densities**

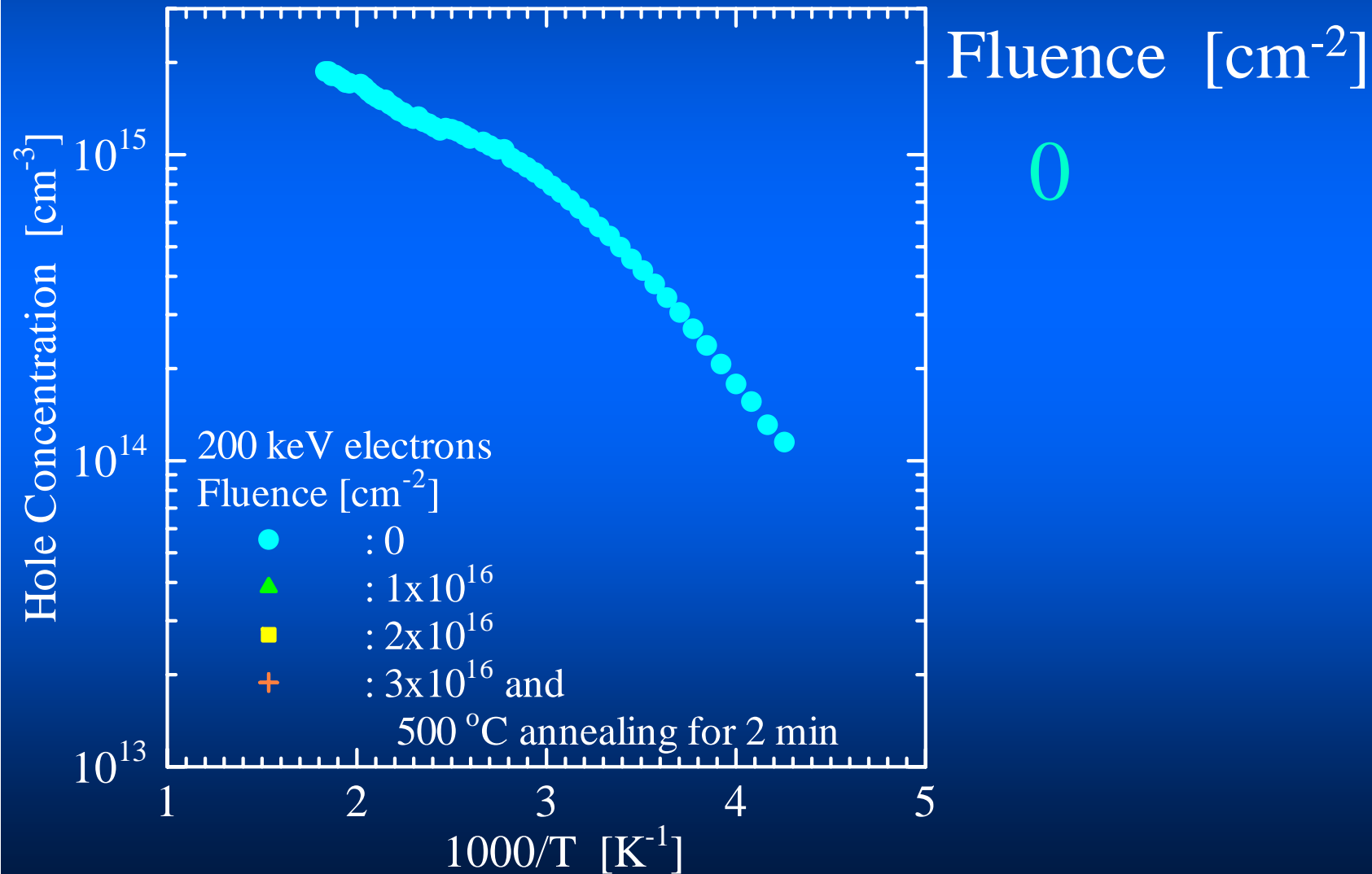
3. What is the origin of the deep acceptor?

## Sample Configuration

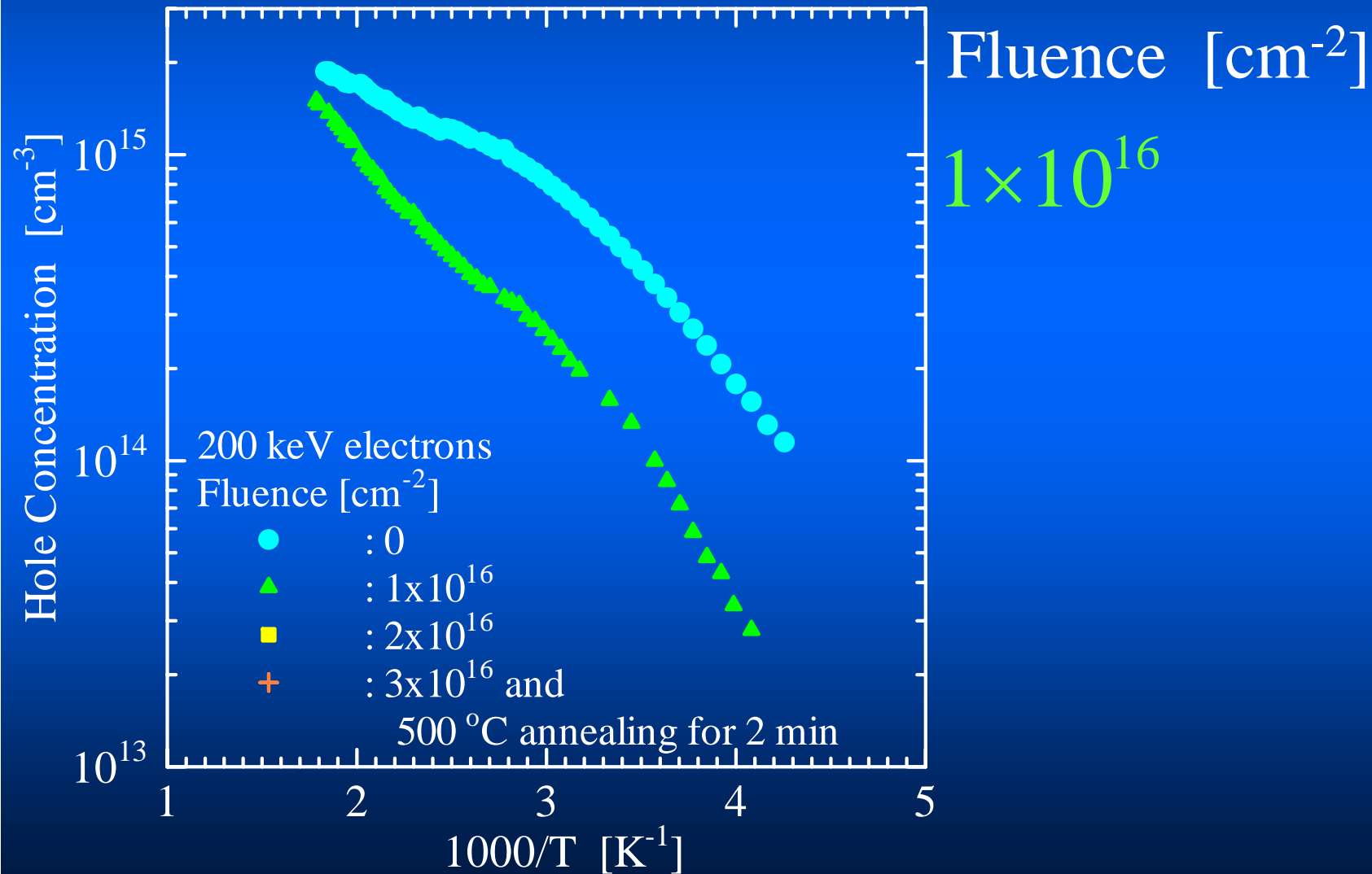
Investigation of acceptors and defects in **Al-doped 6H-SiC** from  $p(T)$  obtained by Hall-effect measurements



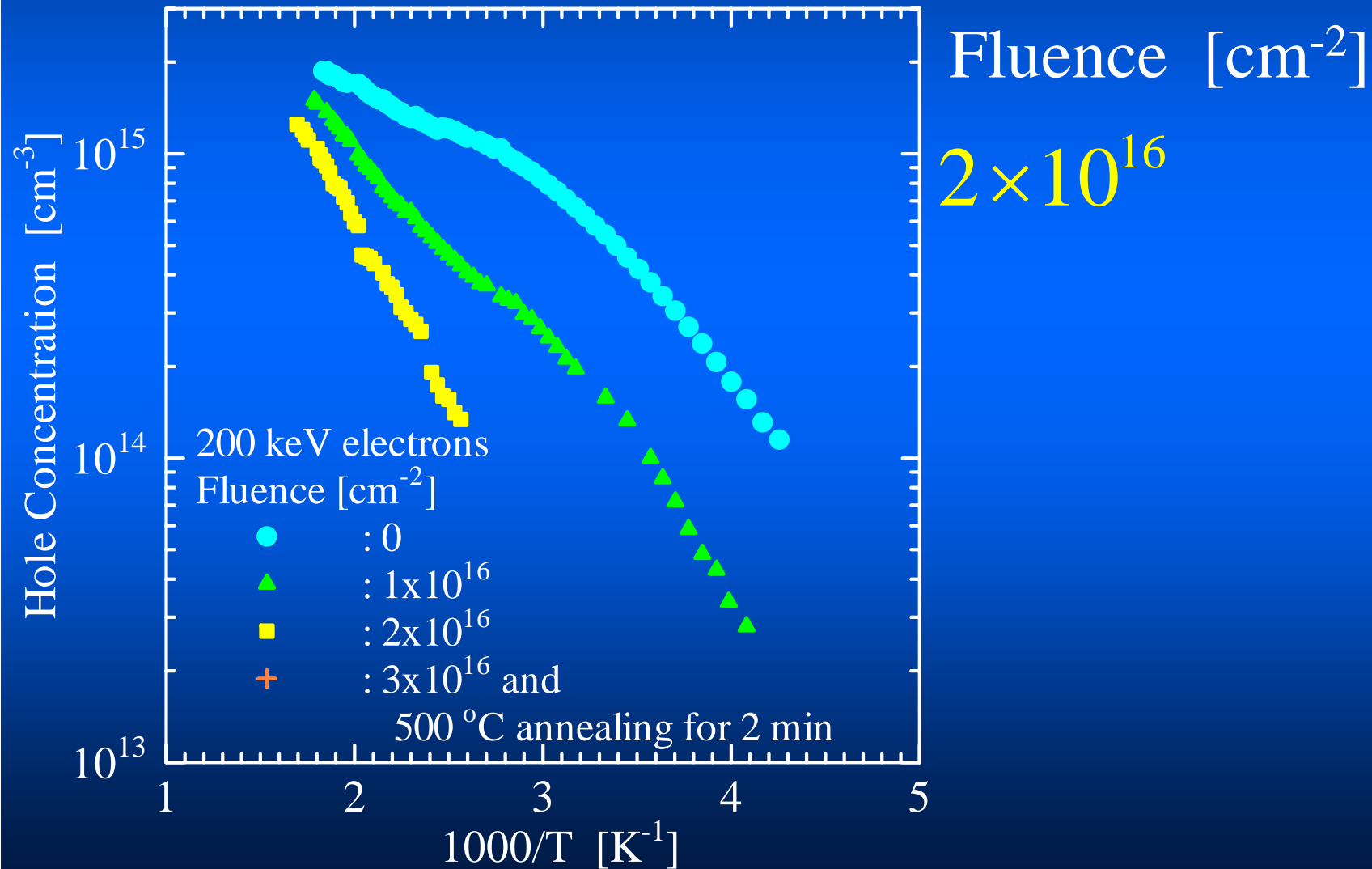
# Reduction in $p(T)$ in Al-doped p-type 6H-SiC by 200 keV electron irradiation



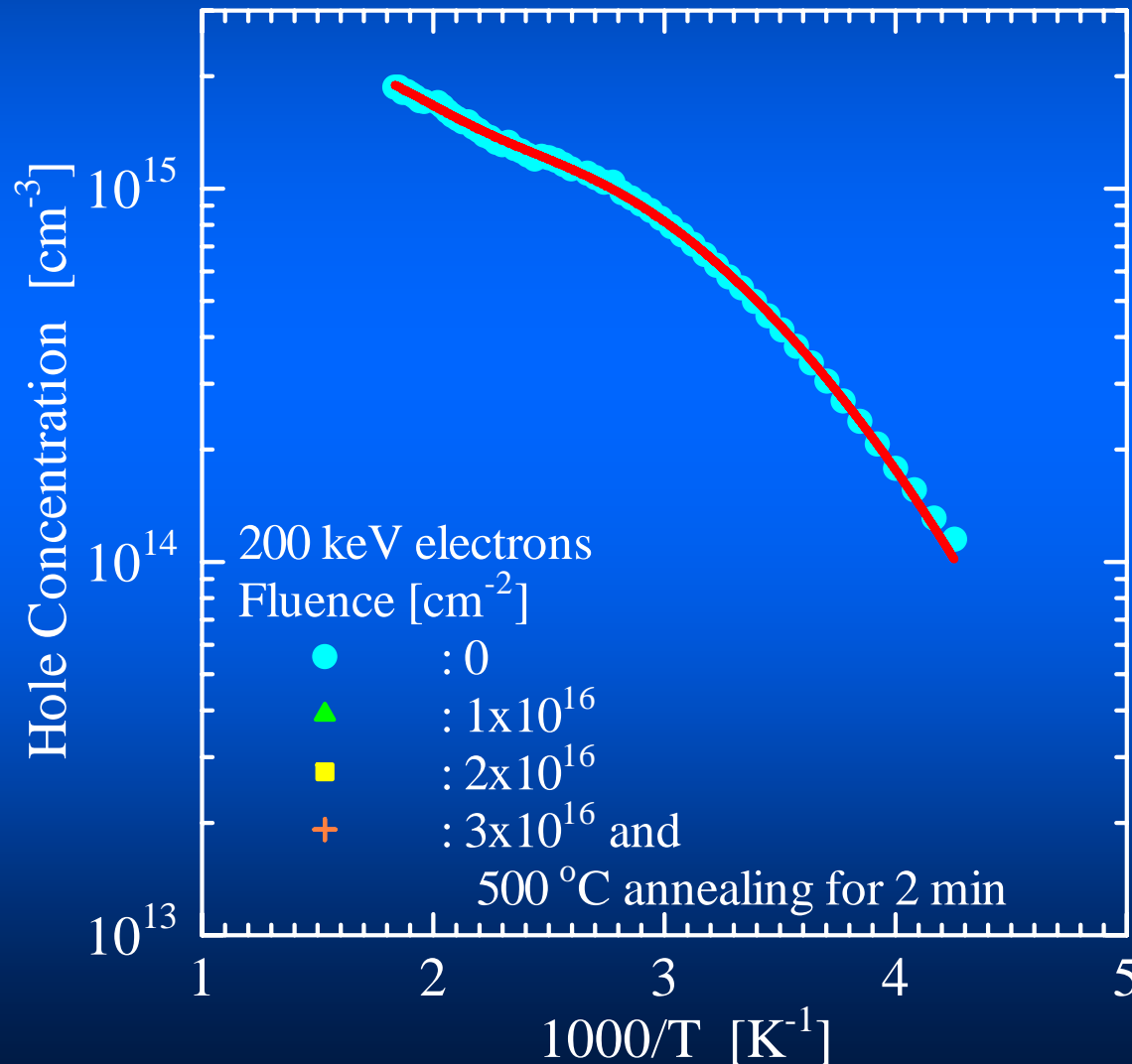
# Reduction in p(T) in Al-doped p-type 6H-SiC by 200 keV electron irradiation



# Reduction in p(T) in Al-doped p-type 6H-SiC by 200 keV electron irradiation



# Reduction in p(T) in Al-doped p-type 6H-SiC by 200 keV electron irradiation



Fluence [cm<sup>-2</sup>]

0

**Simulation Result**

$$N_{Al} = 1.6 \times 10^{15} \text{ cm}^{-3}$$

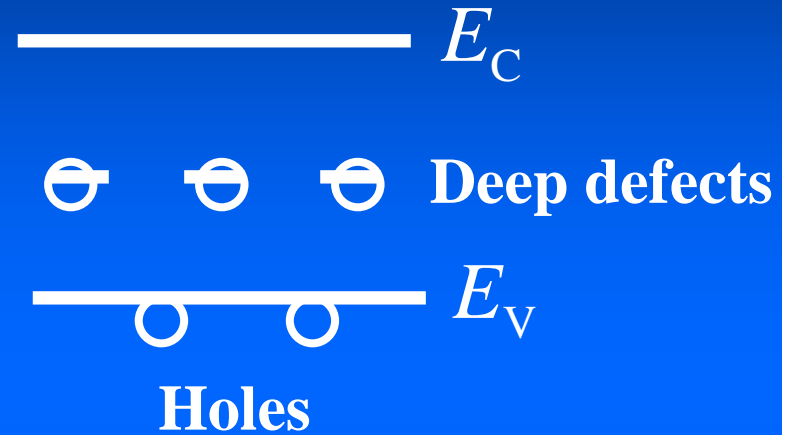
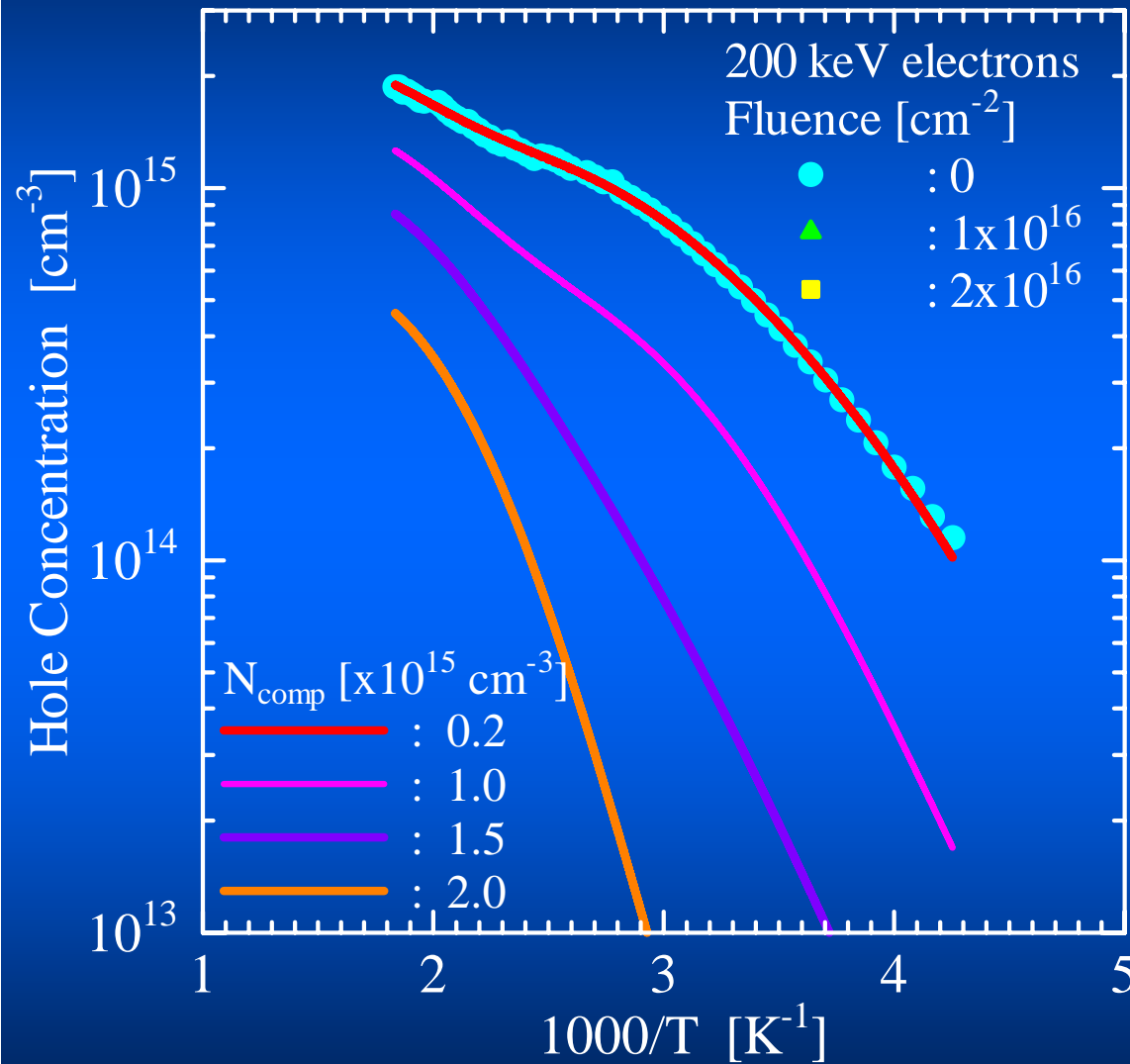
$$E_{Al} = E_V + 0.24 \text{ eV}$$

$$N_{DA} = 1.0 \times 10^{15} \text{ cm}^{-3}$$

$$E_{DA} = E_V + 0.41 \text{ eV}$$

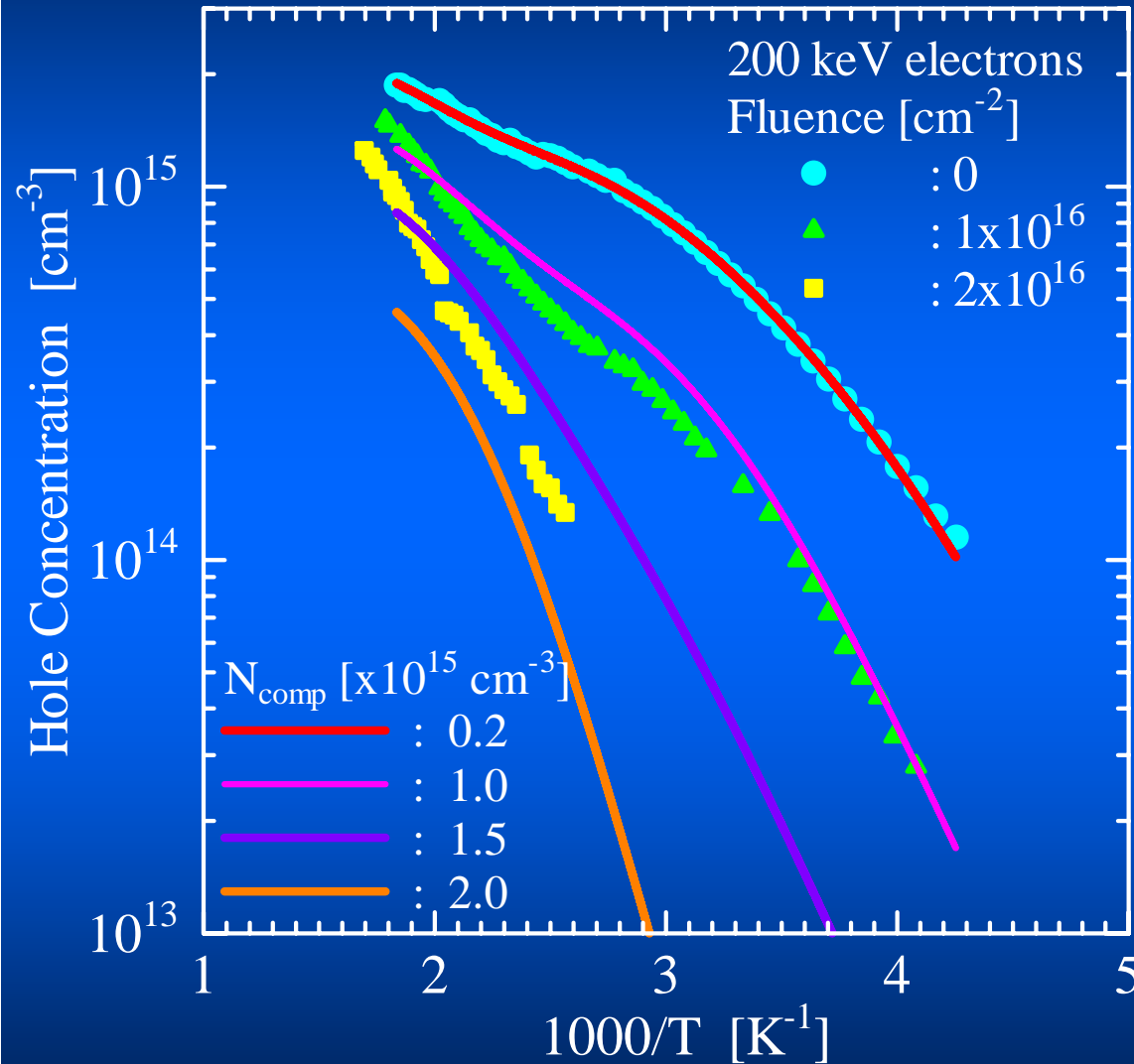
$$N_{comp} = 2.6 \times 10^{14} \text{ cm}^{-3}$$

p(T) influenced by compensating density

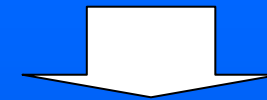


Using densities and energy levels of two types of acceptors in unirradiated sample, p(T) were simulated with different N<sub>comp</sub>.

# p(T) influenced by compensating density



With increasing  $N_{\text{comp}}$ ,  
p(T) values at not  
only low but also high  
temperatures were  
reduced.

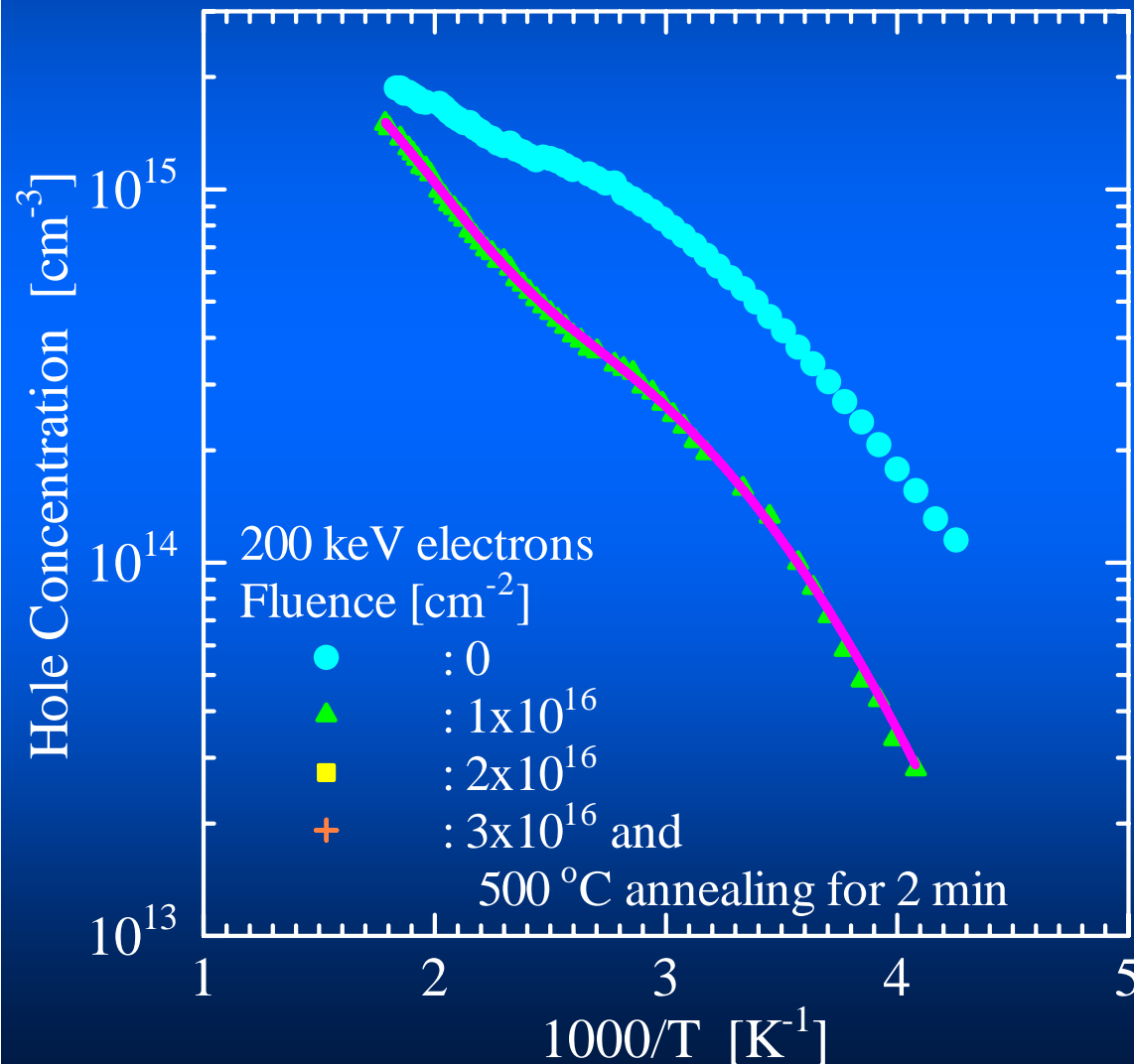


**Experimental p(T)  
values at high  
temperatures were not  
reduced by irradiation  
much than simulated.**

**Acceptor densities should be changed by irradiation.**



# Reduction in p(T) in Al-doped p-type 6H-SiC by 200 keV electron irradiation



Fluence [cm<sup>-2</sup>]

$1 \times 10^{16}$

Simulation Result

$$N_{\text{Al}} = 5.6 \times 10^{14} \text{ cm}^{-3}$$

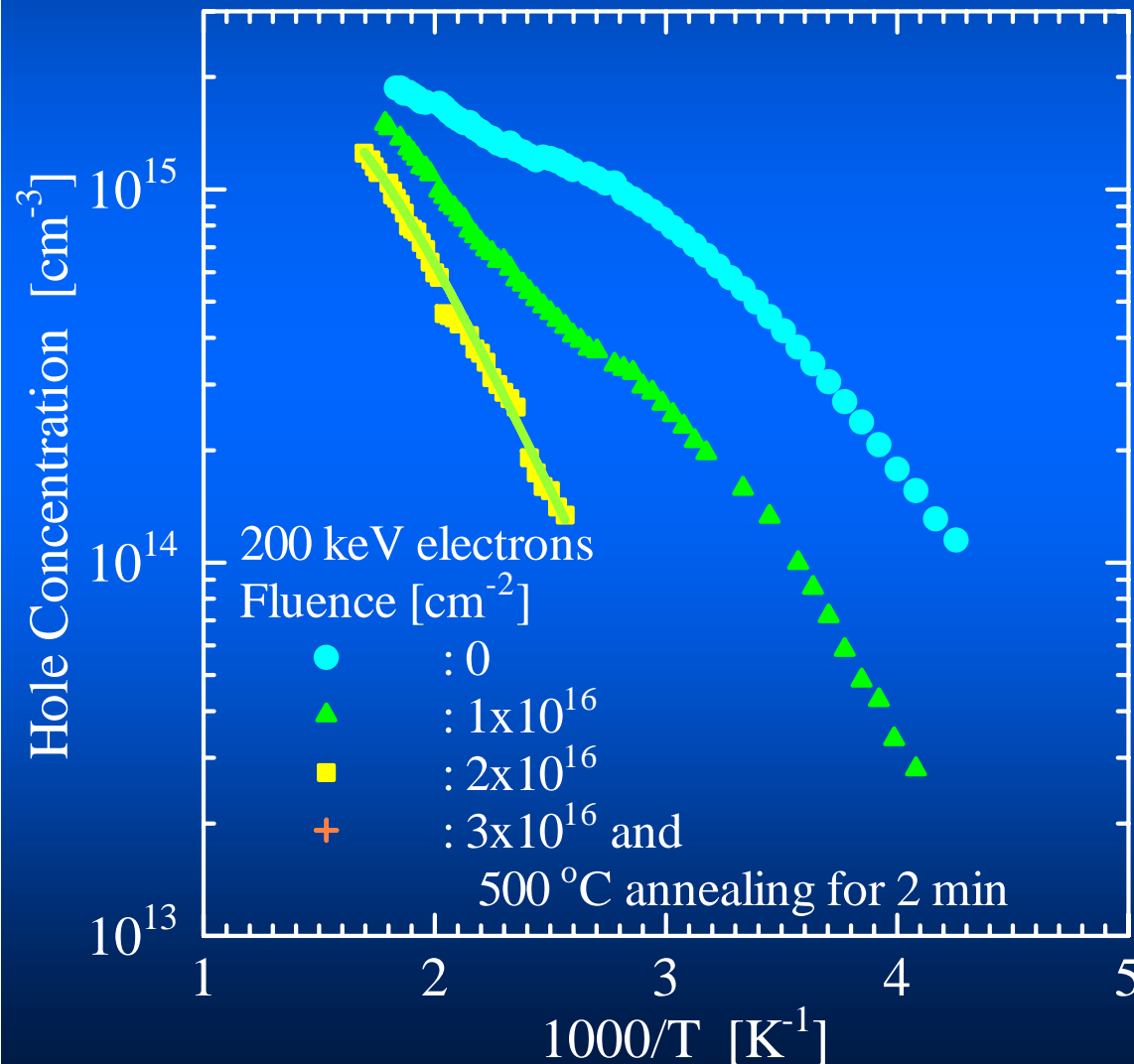
$$E_{\text{Al}} = E_{\text{V}} + 0.27 \text{ eV}$$

$$N_{\text{DA}} = 2.2 \times 10^{15} \text{ cm}^{-3}$$

$$E_{\text{DA}} = E_{\text{V}} + 0.44 \text{ eV}$$

$$N_{\text{comp}} = 1.1 \times 10^{14} \text{ cm}^{-3}$$

# Reduction in p(T) in Al-doped p-type 6H-SiC by 200 keV electron irradiation



Fluence [ $\text{cm}^{-2}$ ]

$$2 \times 10^{16}$$

Simulation Result

$$N_{\text{Al}} < 1 \times 10^{14} \text{ cm}^{-3} \text{ cm}^{-3}$$

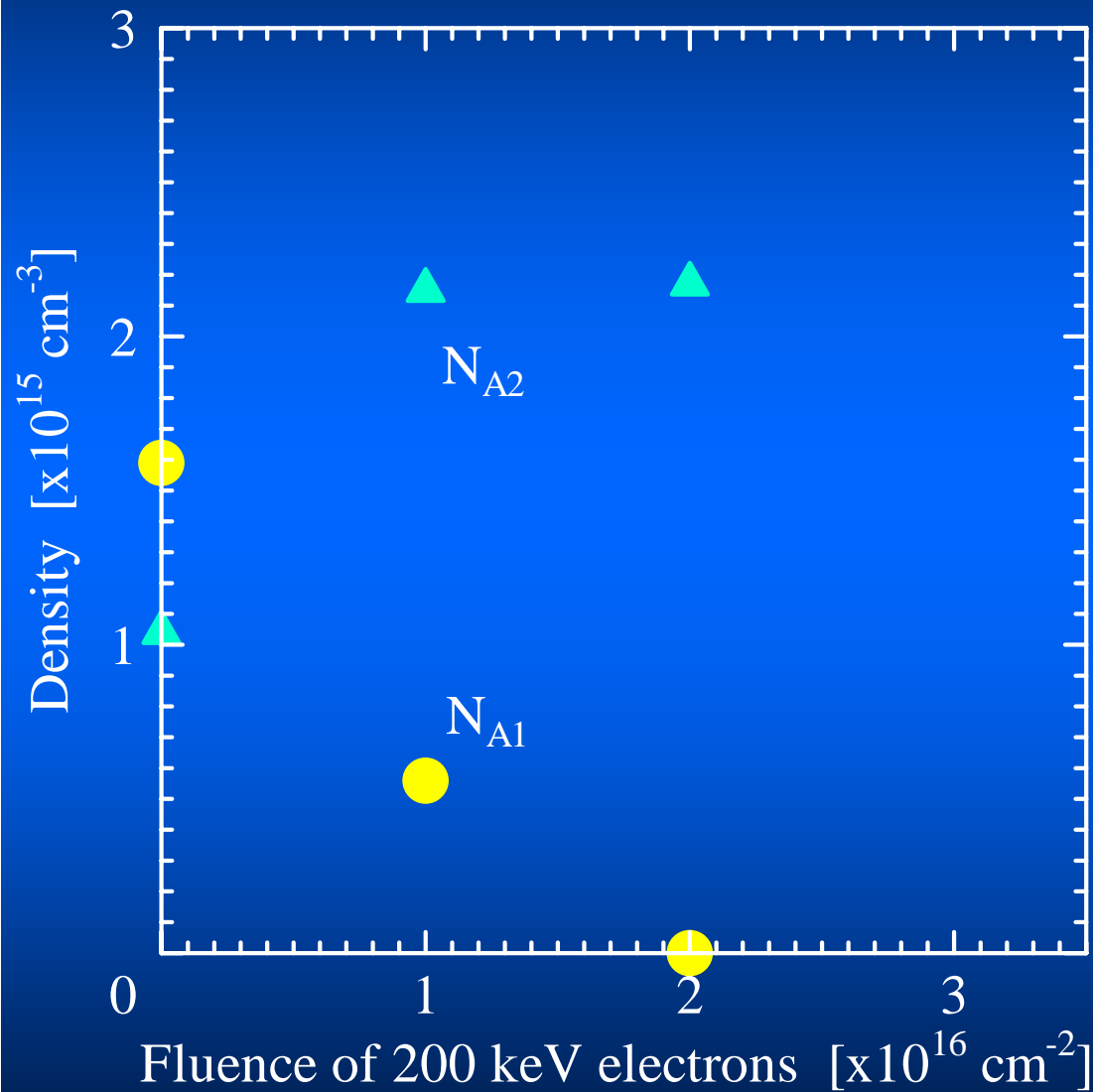
$$E_{\text{Al}} = E_{\text{V}} + 0.24 \text{ eV}$$

$$N_{\text{DA}} = 2.2 \times 10^{15} \text{ cm}^{-3}$$

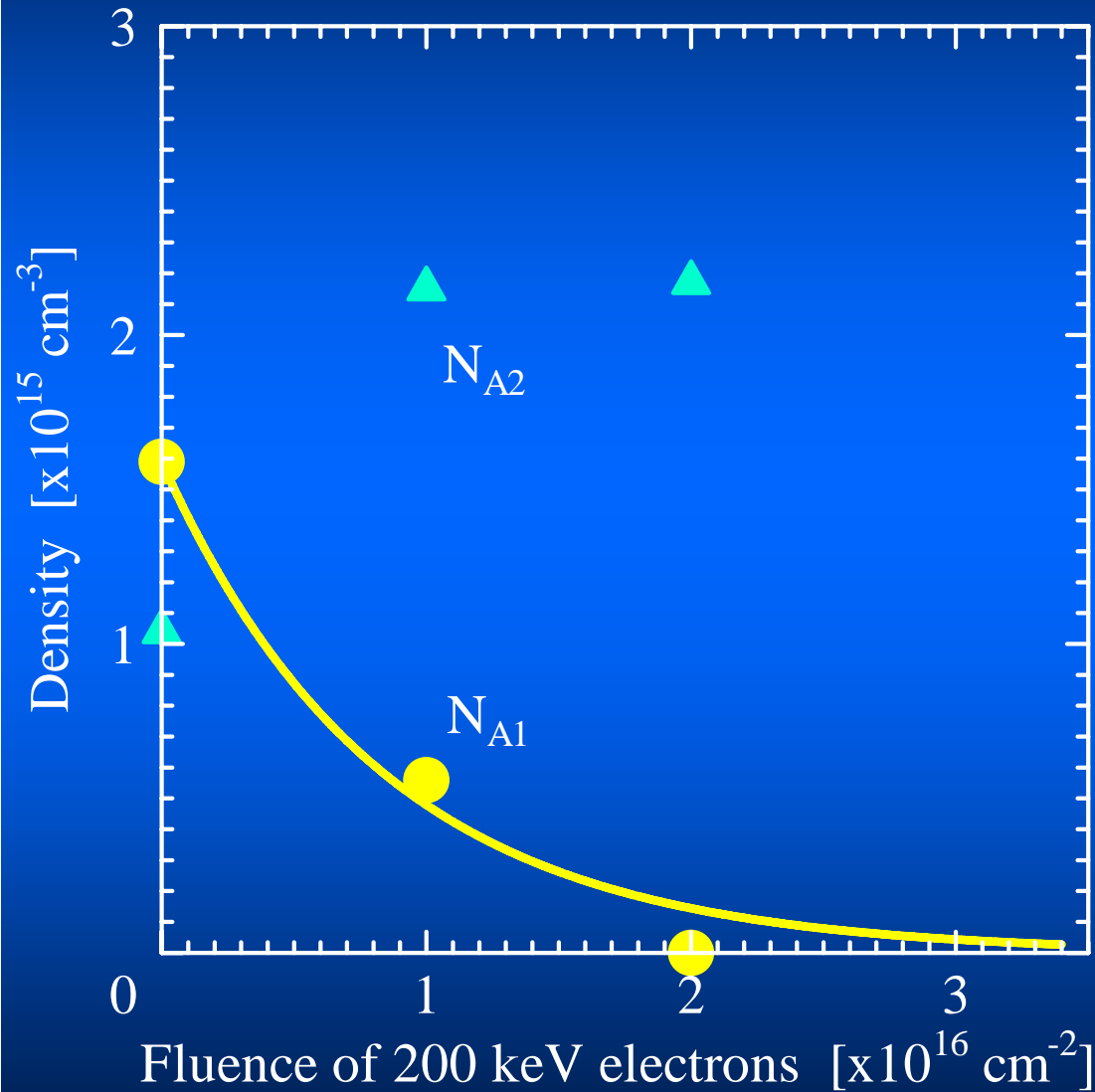
$$E_{\text{DA}} = E_{\text{V}} + 0.47 \text{ eV}$$

$$N_{\text{comp}} = 1.1 \times 10^{12} \text{ cm}^{-3}$$

# Fluence Dependence of $N_{A1}$ and $N_{DA}$ in 6H-SiC



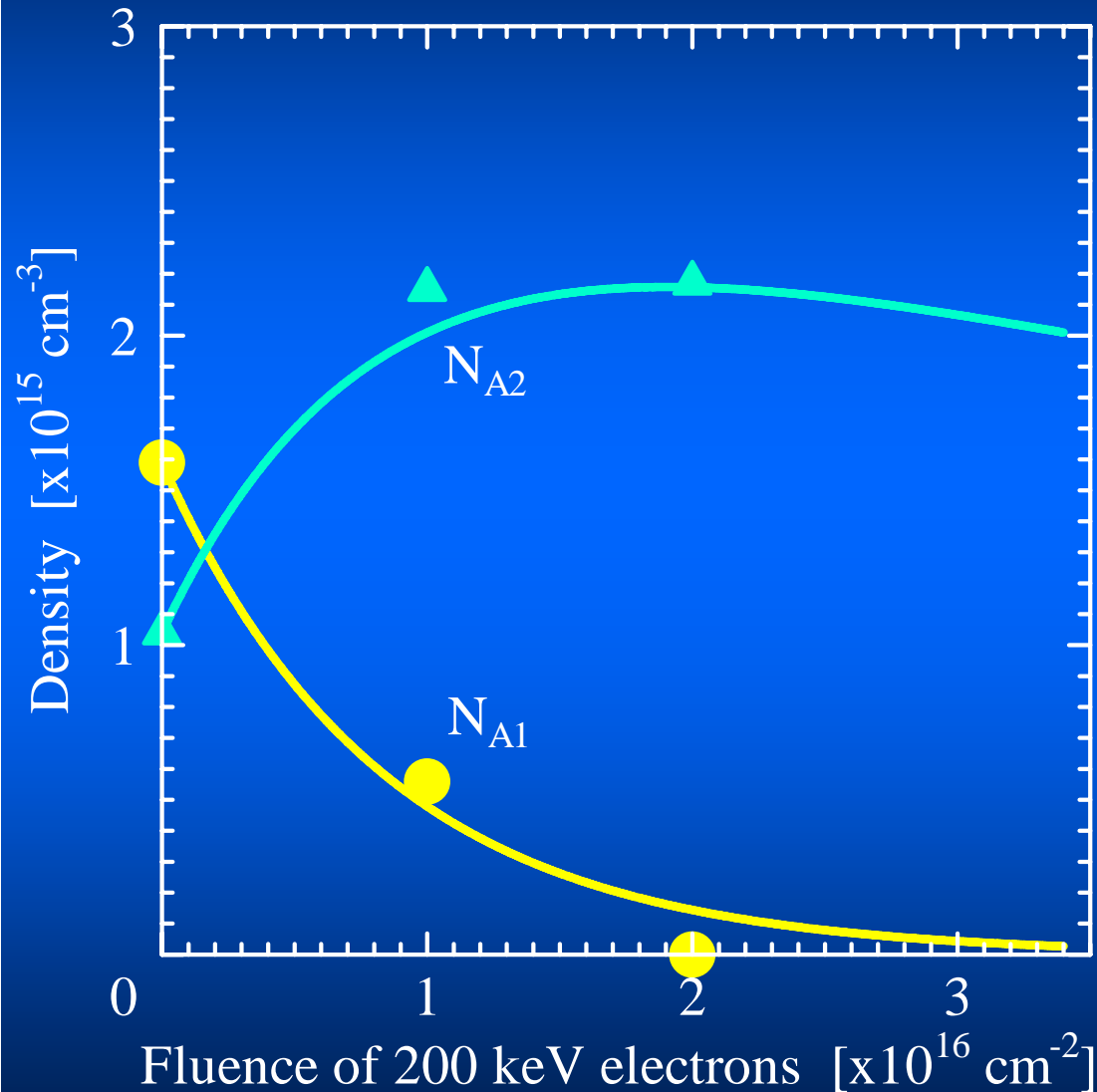
# Fluence Dependence of $N_{Al}$ and $N_{DA}$ in 6H-SiC



$$\frac{dN_{Al}(\Phi)}{d\Phi} = -\kappa_{Al} N_{Al}(\Phi)$$

$$\kappa_{Al} = 1 \times 10^{-16} \text{ cm}^2$$

## Fluence Dependence of $N_{Al}$ and $N_{DA}$ in 6H-SiC



$$\frac{dN_{Al}(\Phi)}{d\Phi} = -\kappa_{Al}N_{Al}(\Phi)$$

$$\kappa_{Al} = 4.4 \times 10^{-17} \text{ cm}^2$$

$$\frac{dN_{DA}(\Phi)}{d\Phi} = -\frac{dN_{Al}(\Phi)}{d\Phi} - \kappa_{DA}N_{DA}(\Phi)$$

$$\kappa_{DA} = 9 \times 10^{-18} \text{ cm}^2$$

## Comparison of removal cross sections for Al-doped 4H-SiC and 6H-SiC by 200 keV electron irradiation

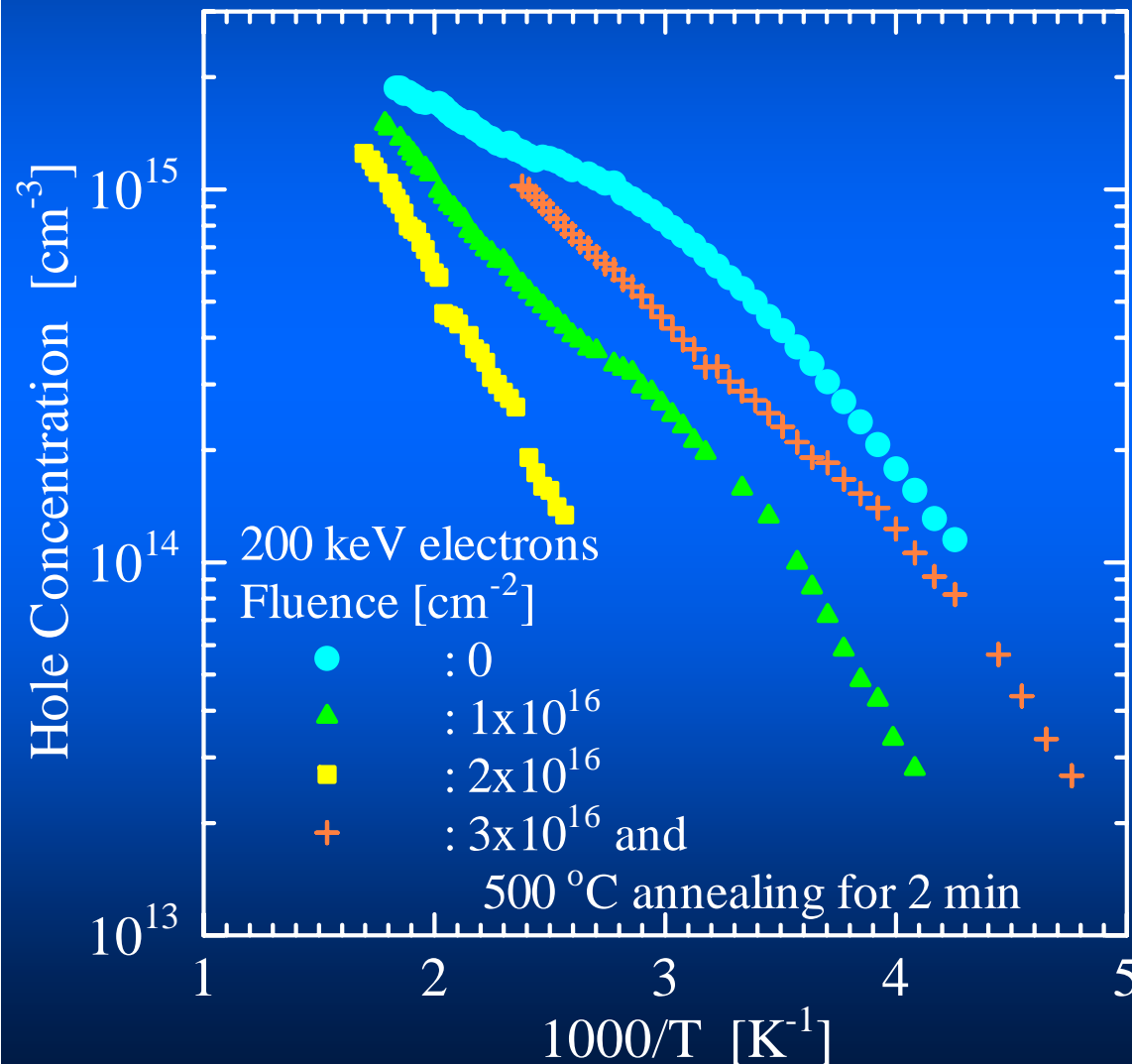
	Al-doped 6H-SiC	Al-doped 4H-SiC
$\kappa_{\text{Al}}$ [cm <sup>2</sup> ]	$1 \times 10^{-16}$	$4.4 \times 10^{-17}$
$\kappa_{\text{DA}}$ [cm <sup>2</sup> ]	$9 \times 10^{-18}$	$1.0 \times 10^{-17}$

$\kappa_{\text{Al}}$  for 6H-SiC is larger than that for 4H-SiC



**Al-doped 6H-SiC is radiation-resistant less than  
Al-doped 4H-SiC**

# Reduction in p(T) in Al-doped p-type 6H-SiC by 200 keV electron irradiation



Fluence [ $\text{cm}^{-2}$ ]

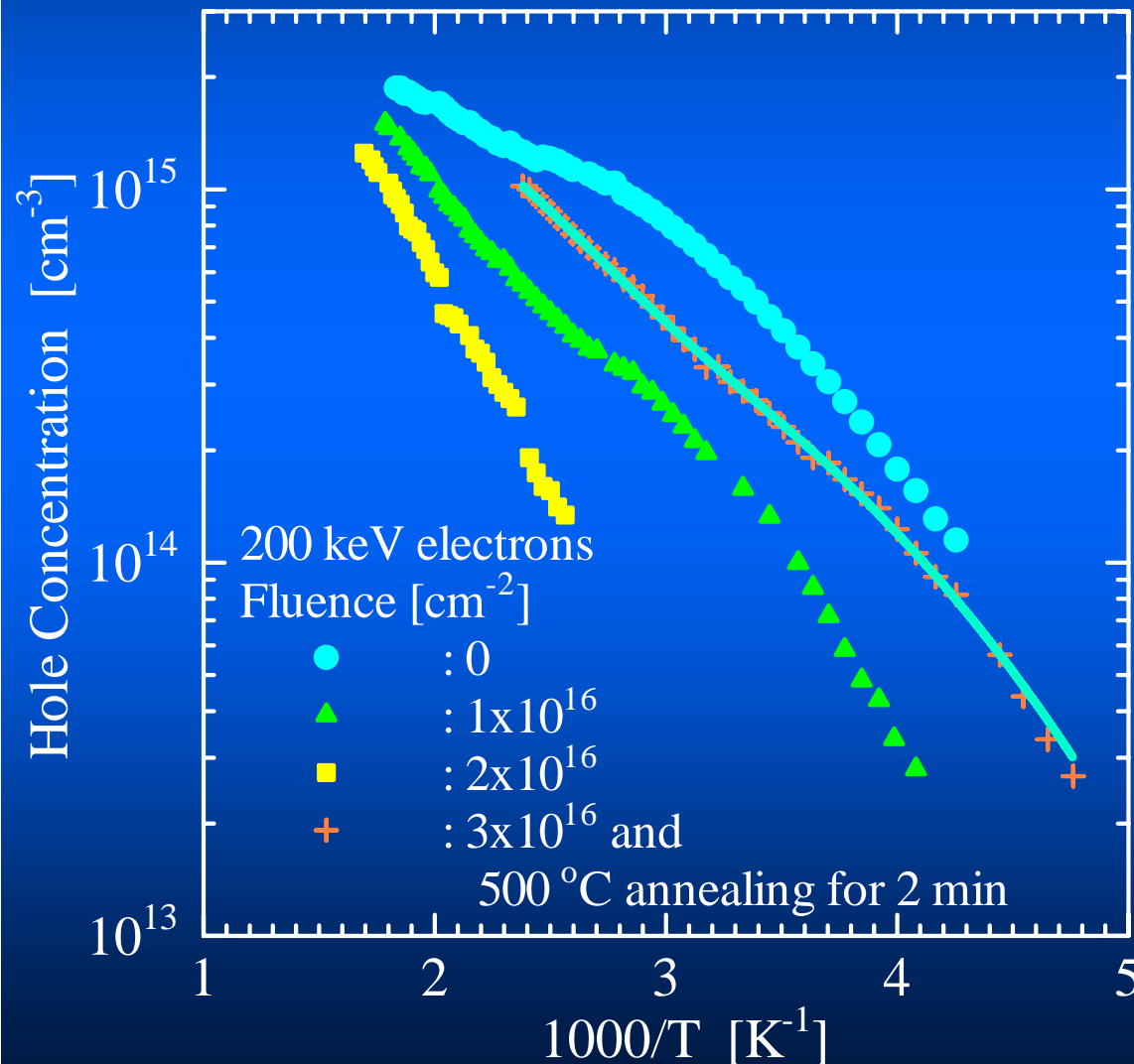
$3 \times 10^{16}$

The sample irradiated by  $3 \times 10^{16} \text{ cm}^{-2}$  fluence could not be measured.



Therefore, the sample was annealed at 500 °C for 2 min.

# Recovery in p(T) in Al-doped p-type 6H-SiC by 500 °C annealing



Fluence:  $3 \times 10^{16} \text{ cm}^{-2}$   
Annealing: 500 °C  
2 min

## Simulation Result

$$N_{\text{Al}} = 4.0 \times 10^{15} \text{ cm}^{-3}$$

$$E_{\text{Al}} = E_{\text{V}} + 0.23 \text{ eV}$$

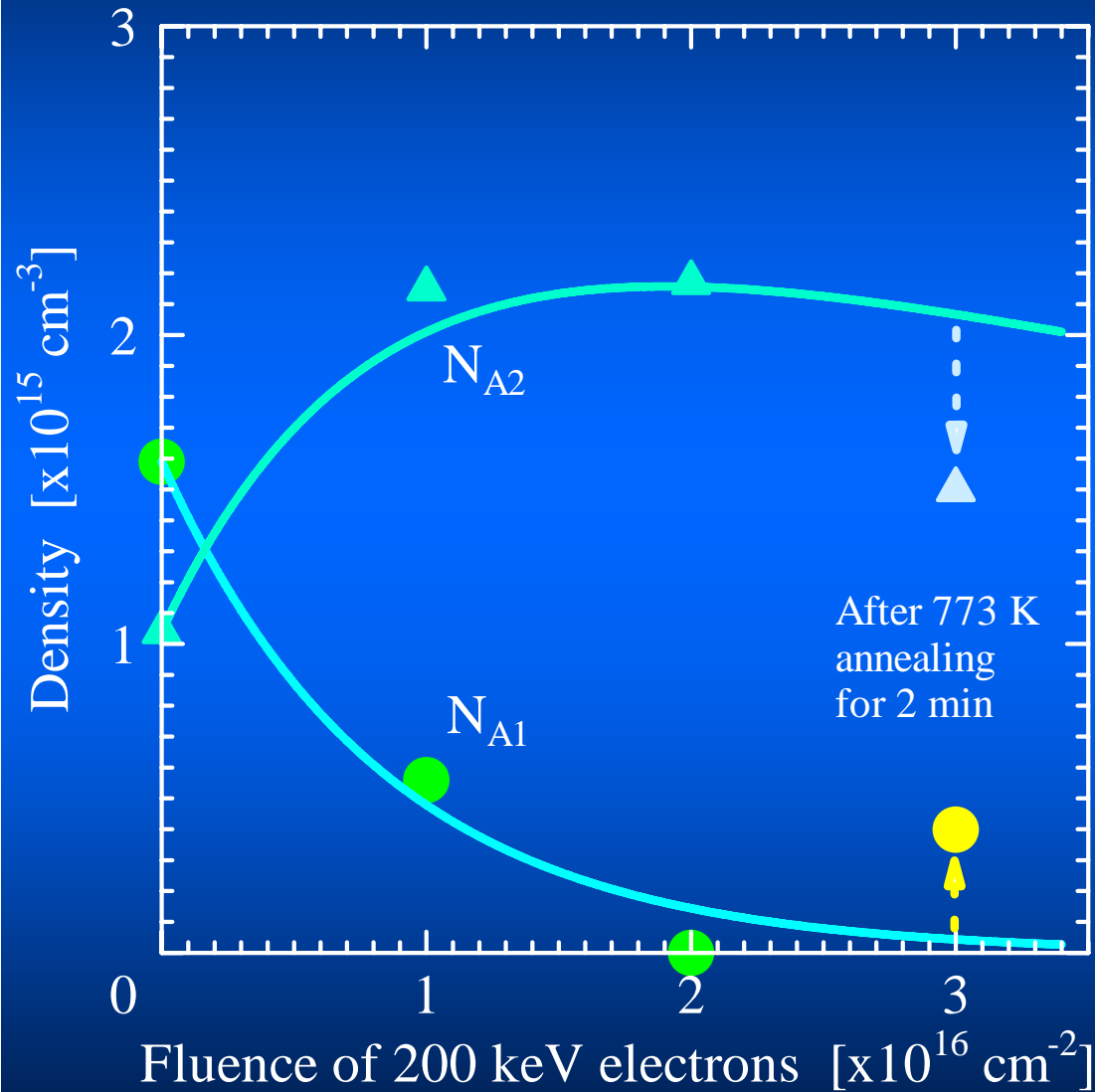
$$N_{\text{DA}} = 1.5 \times 10^{15} \text{ cm}^{-3}$$

$$E_{\text{DA}} = E_{\text{V}} + 0.34 \text{ eV}$$

$$N_{\text{comp}} = 6.6 \times 10^{13} \text{ cm}^{-3}$$



# Fluence Dependence of $N_{Al}$ and $N_{DA}$ in 6H-SiC



500 °C Annealing for 2 min

Increment of  $N_{Al}$ :

$$4 \times 10^{14} \text{ cm}^{-3}$$

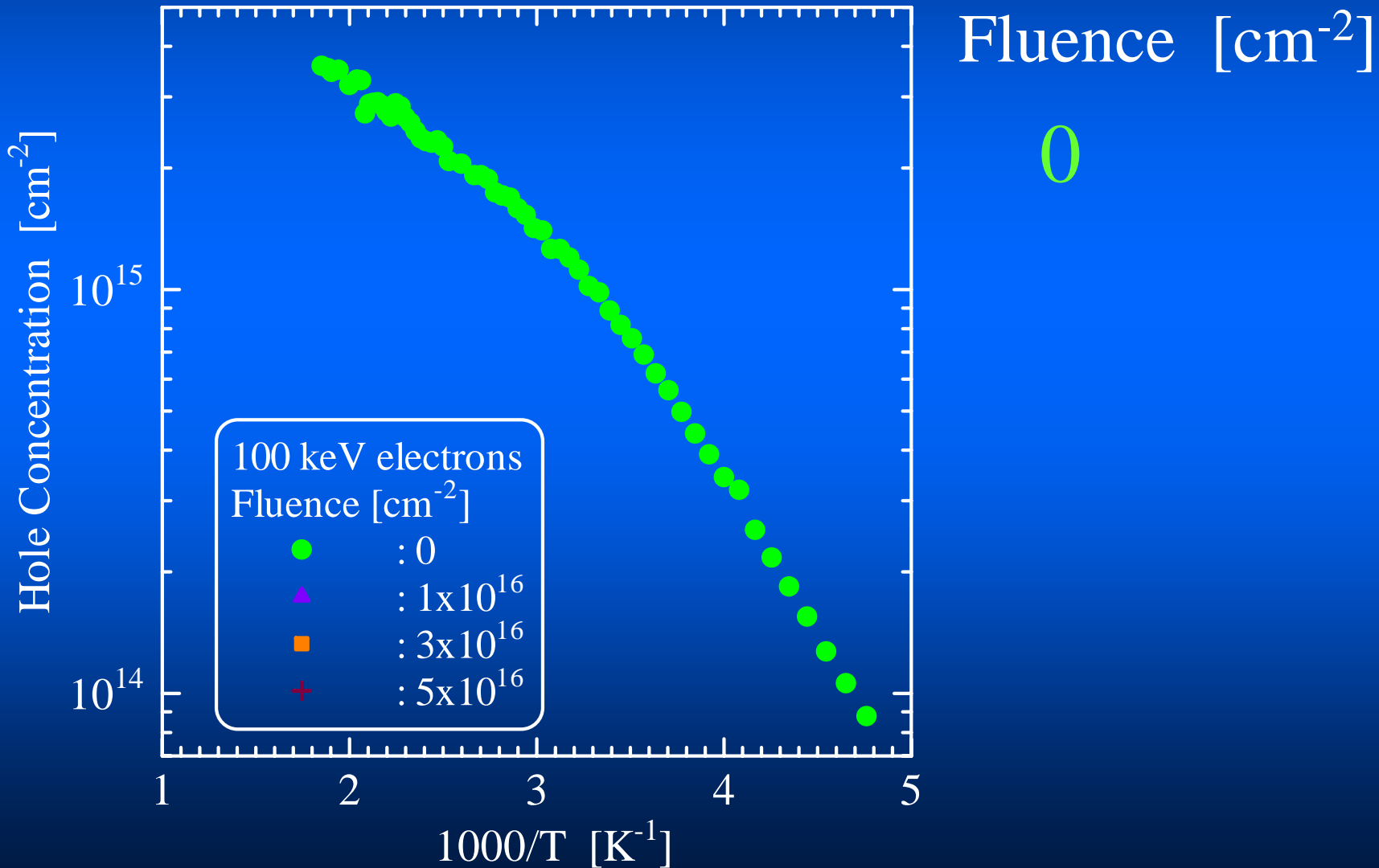


Almost the same

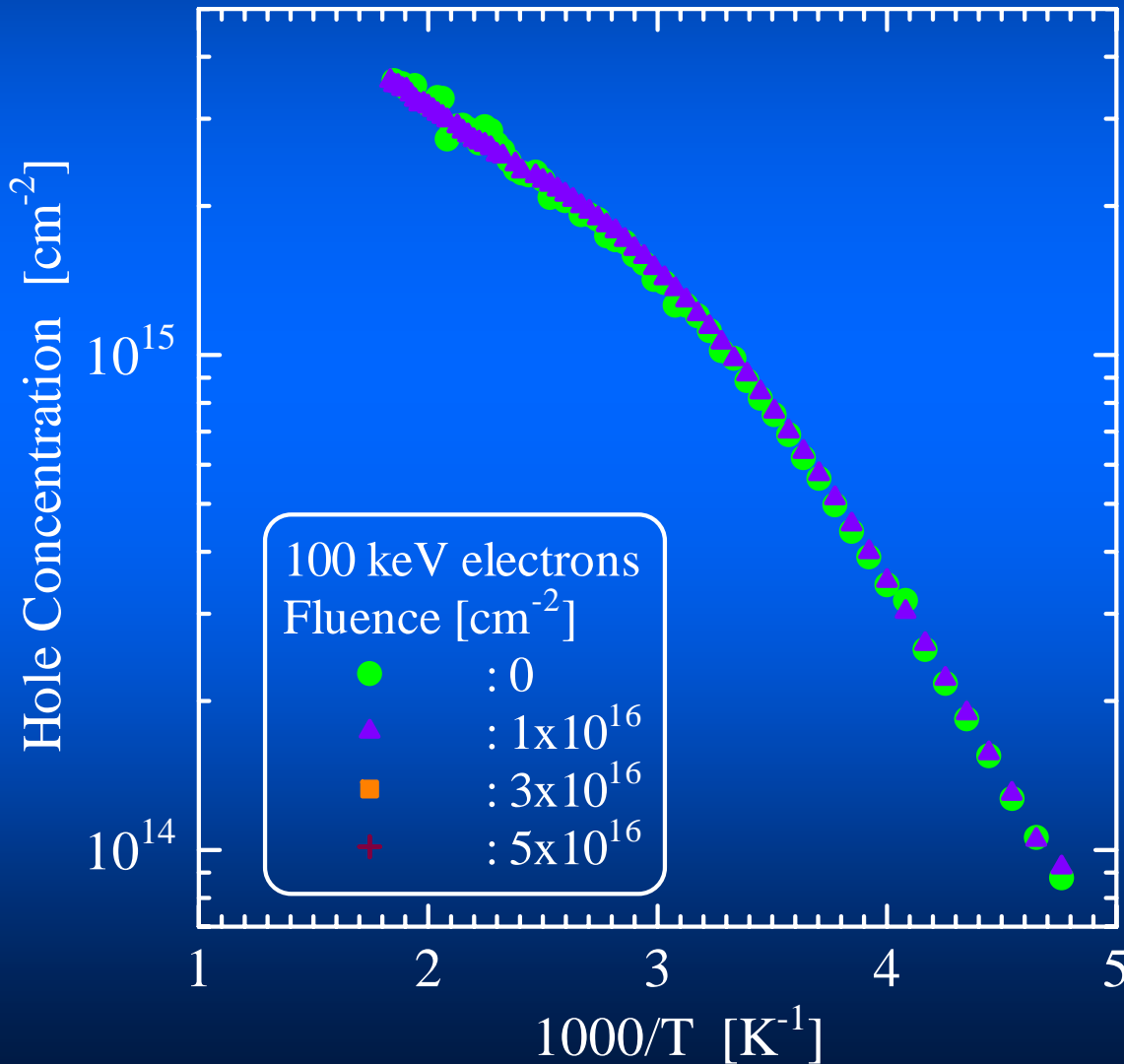
Decrement of  $N_{DA}$ :

$$6 \times 10^{14} \text{ cm}^{-3}$$

# Reduction in $p(T)$ in Al-doped p-type 6H-SiC by **100 keV** electron irradiation



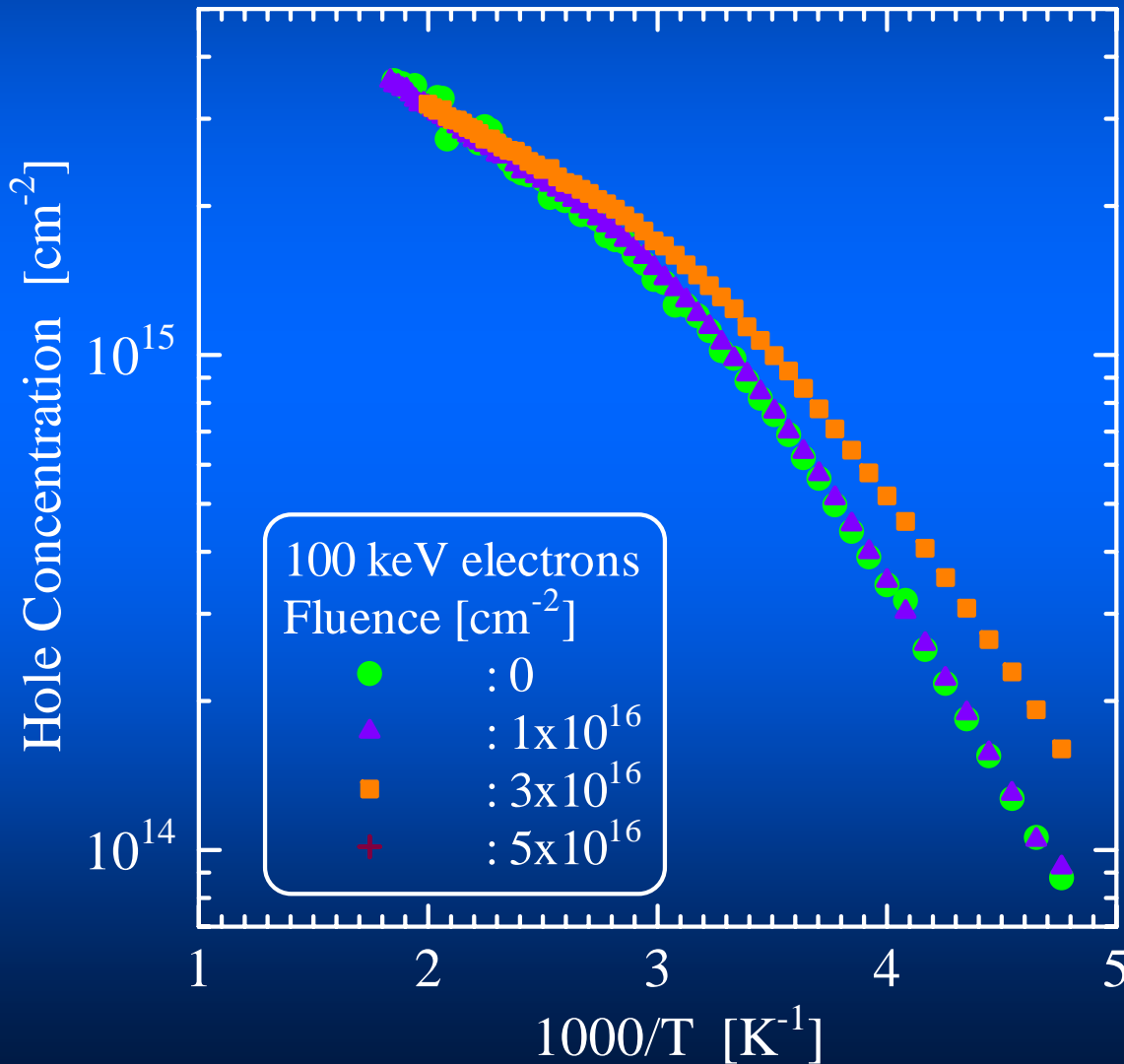
# Reduction in $p(T)$ in Al-doped p-type 6H-SiC by **100 keV** electron irradiation



Fluence [ $\text{cm}^{-2}$ ]

$1 \times 10^{16}$

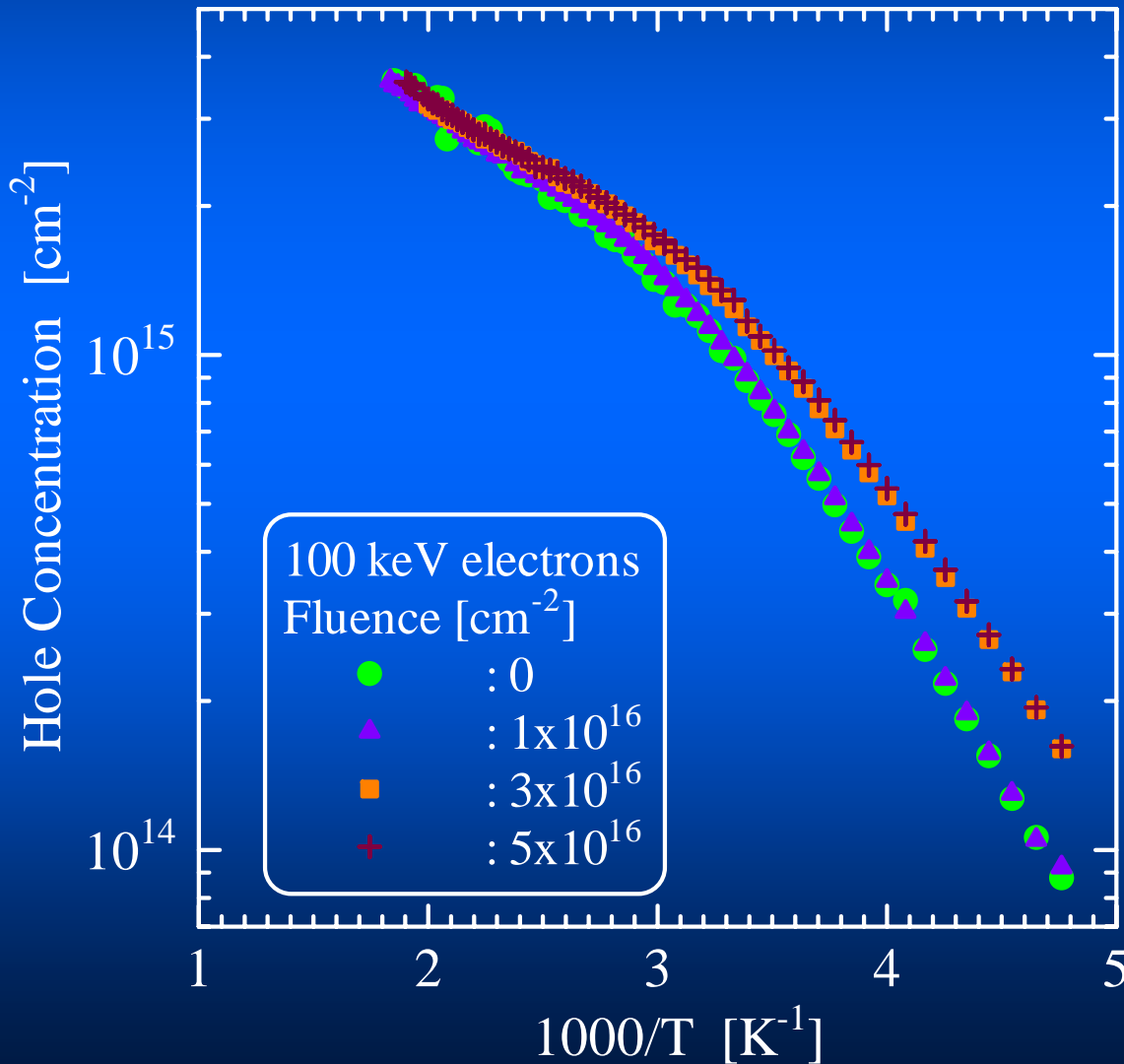
# Reduction in p(T) in Al-doped p-type 6H-SiC by **100 keV** electron irradiation



Fluence [ $\text{cm}^{-2}$ ]

$3 \times 10^{16}$

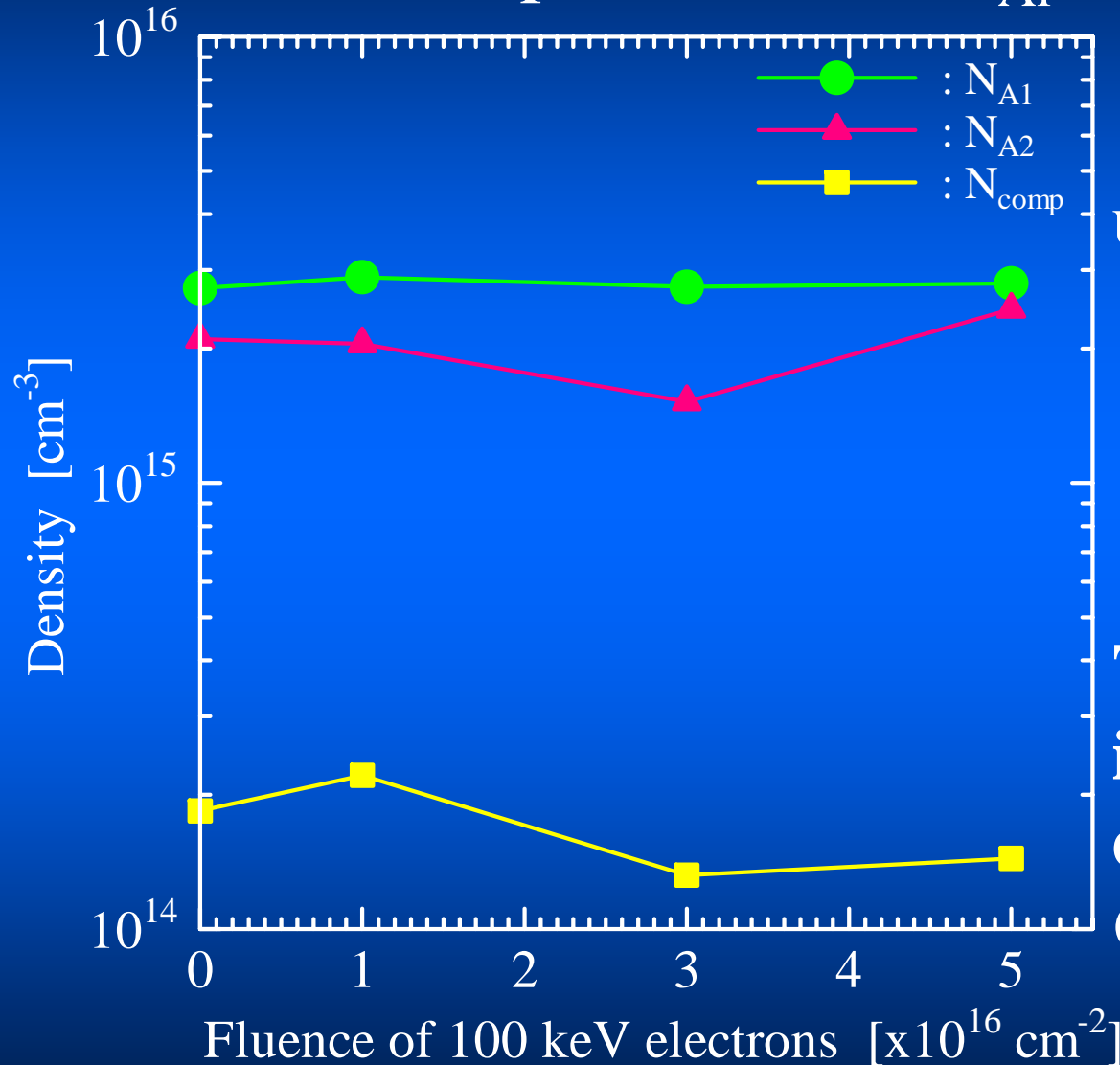
# Reduction in p(T) in Al-doped p-type 6H-SiC by **100 keV** electron irradiation



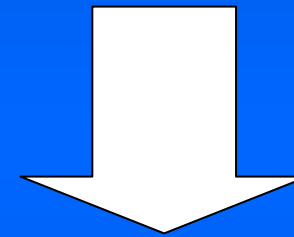
Fluence [ $\text{cm}^{-2}$ ]

$5 \times 10^{16}$

# Fluence Dependence of $N_{Al}$ and $N_{DA}$ in 6H-SiC

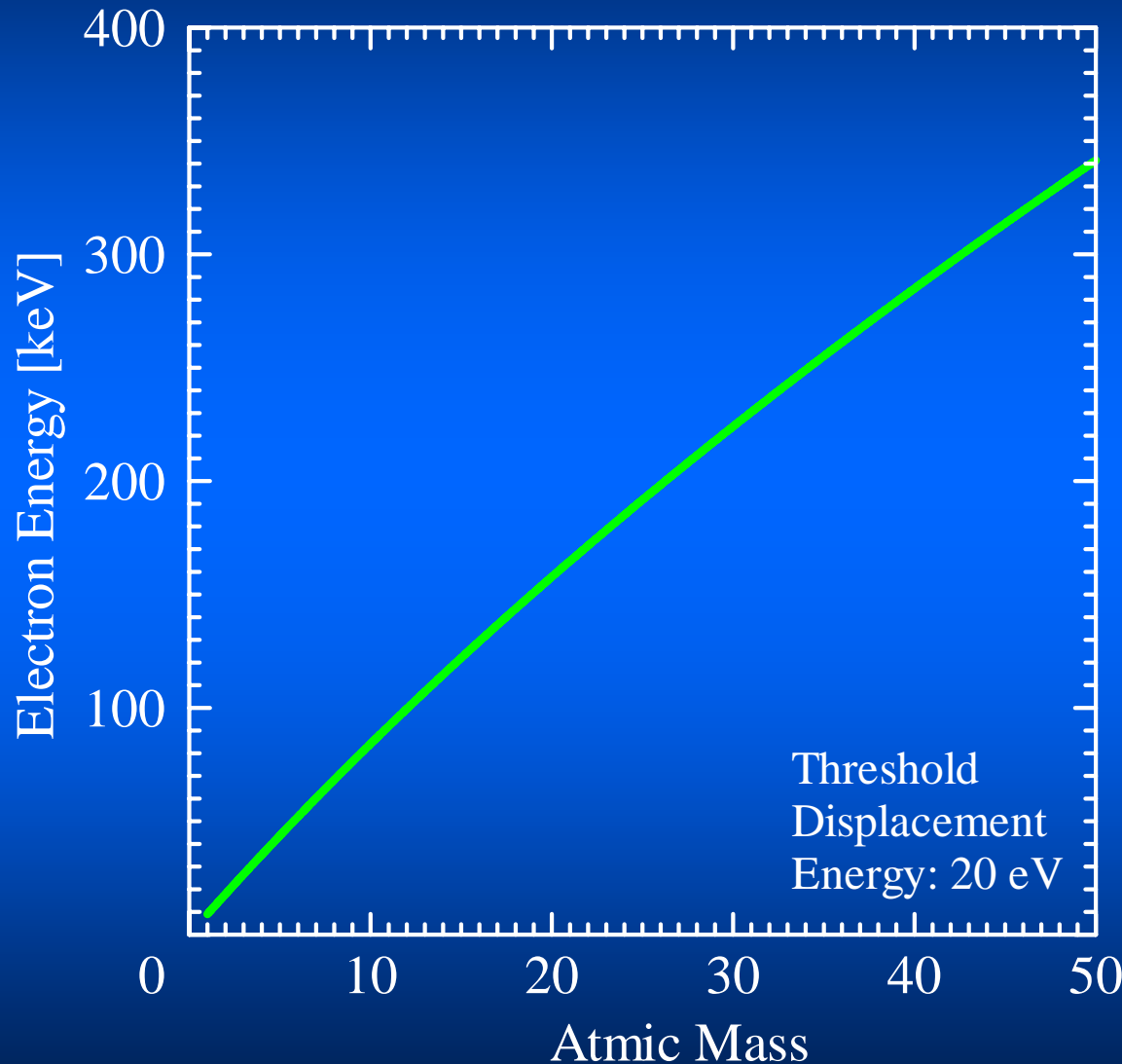


$N_{Al}$  and  $N_{DA}$  are almost unchanged.



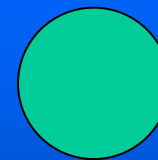
The 100 keV electron irradiation cannot displace substitutional C.

Minimum Electron Energy for displacement of Substitutional Atom



Head-on collision

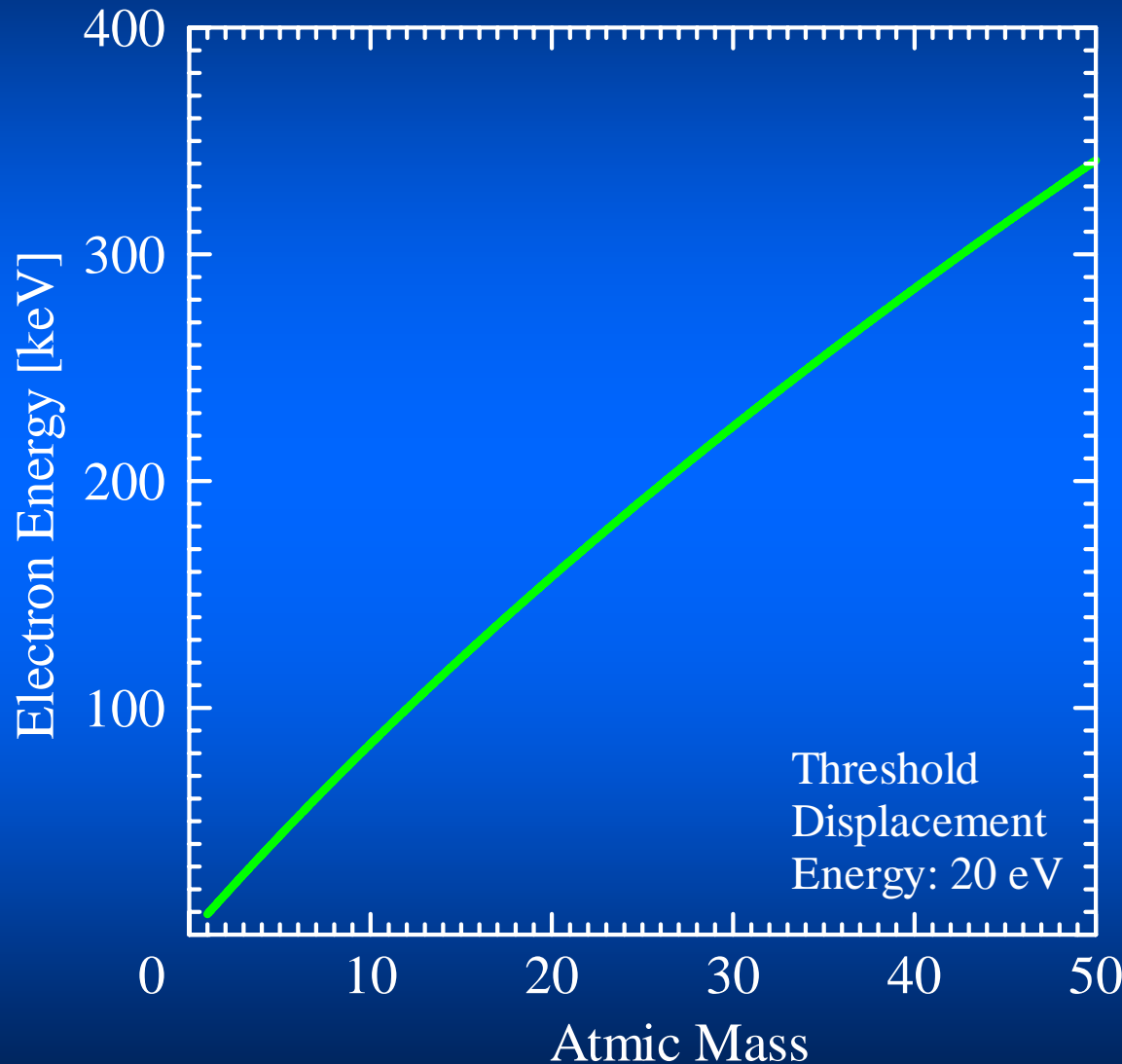
An electron having  $E$  keV



Atom

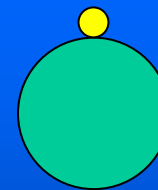
Threshold displacement energy:  $E_d$  eV

Minimum Electron Energy for displacement of Substitutional Atom



Head-on collision

The electron having  $E$  keV loses energy of  $\alpha$  eV

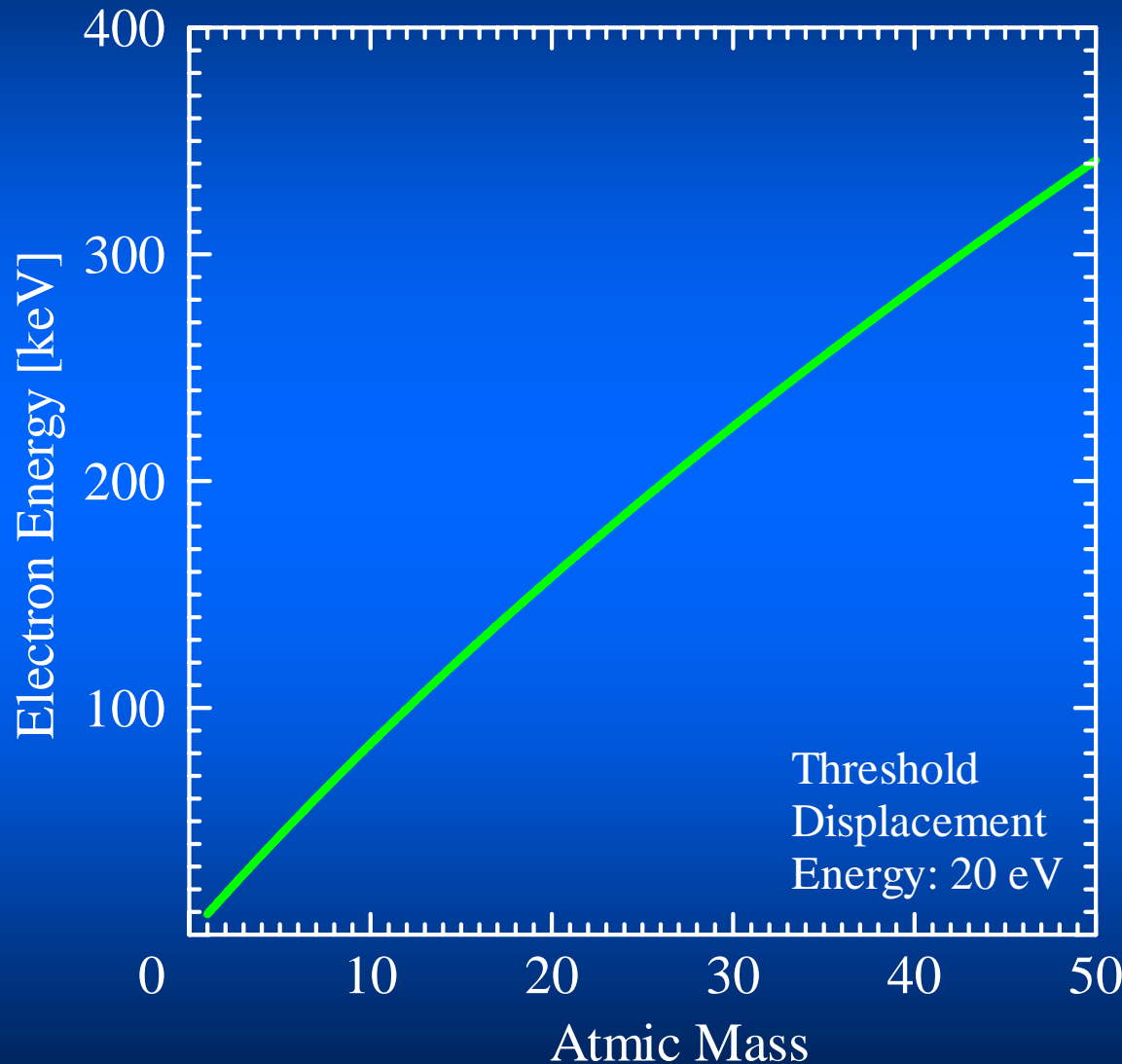


If  $\alpha \geq E_d$

Threshold displacement energy:  $E_d$  eV



Minimum Electron Energy for displacement of Substitutional Atom



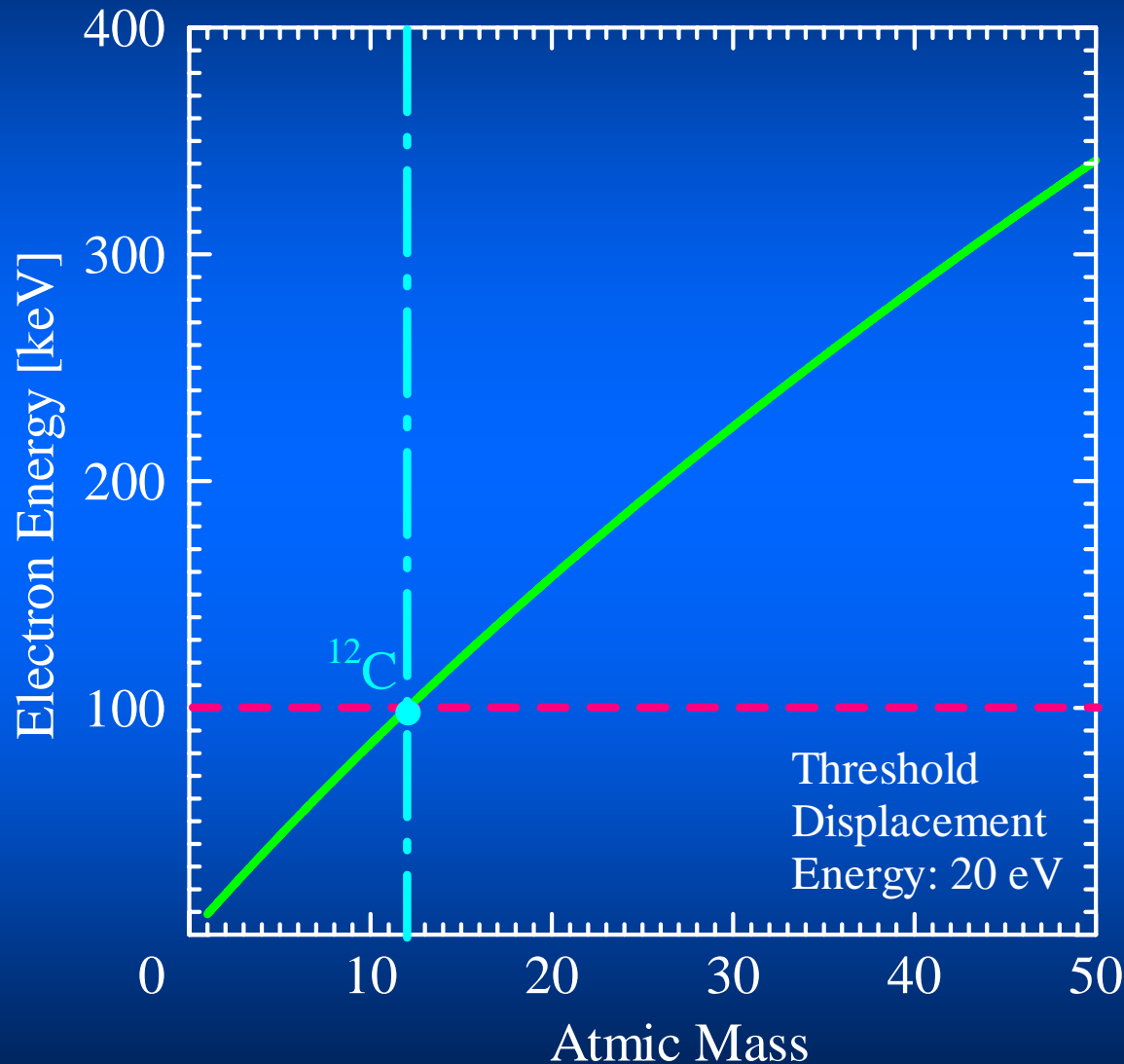
Head-on collision

The electron having  $E$  keV loses energy of  $\alpha$  eV



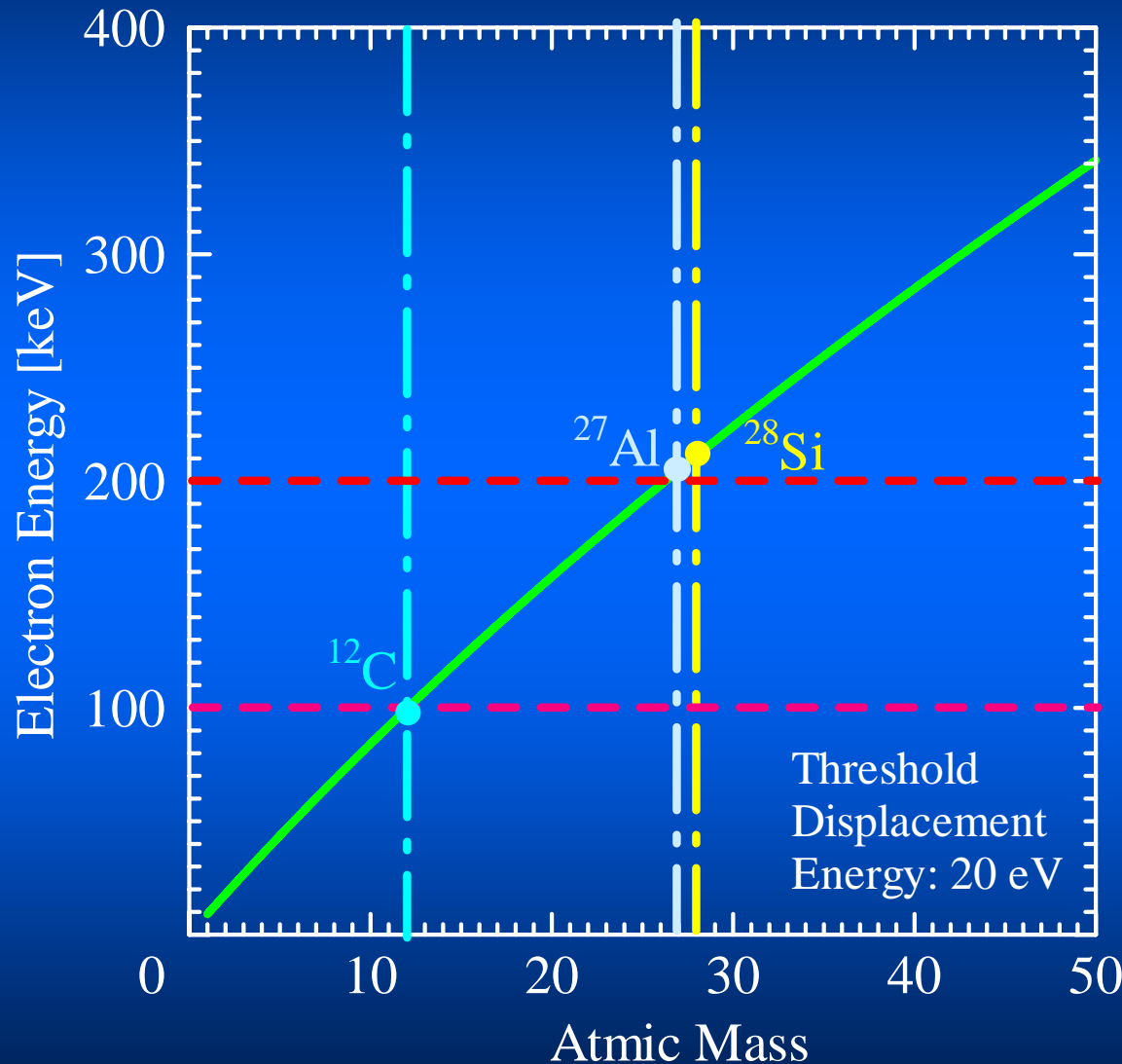
If  $\alpha \geq E_d$  the atom is displaced.

Minimum Electron Energy for displacement of Substitutional Atom

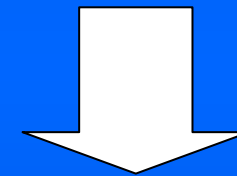


In the case of  $E_d=20$  eV, the substitutional C can be displaced by 100 keV electron irradiation.

Minimum Electron Energy for displacement of Substitutional Atom

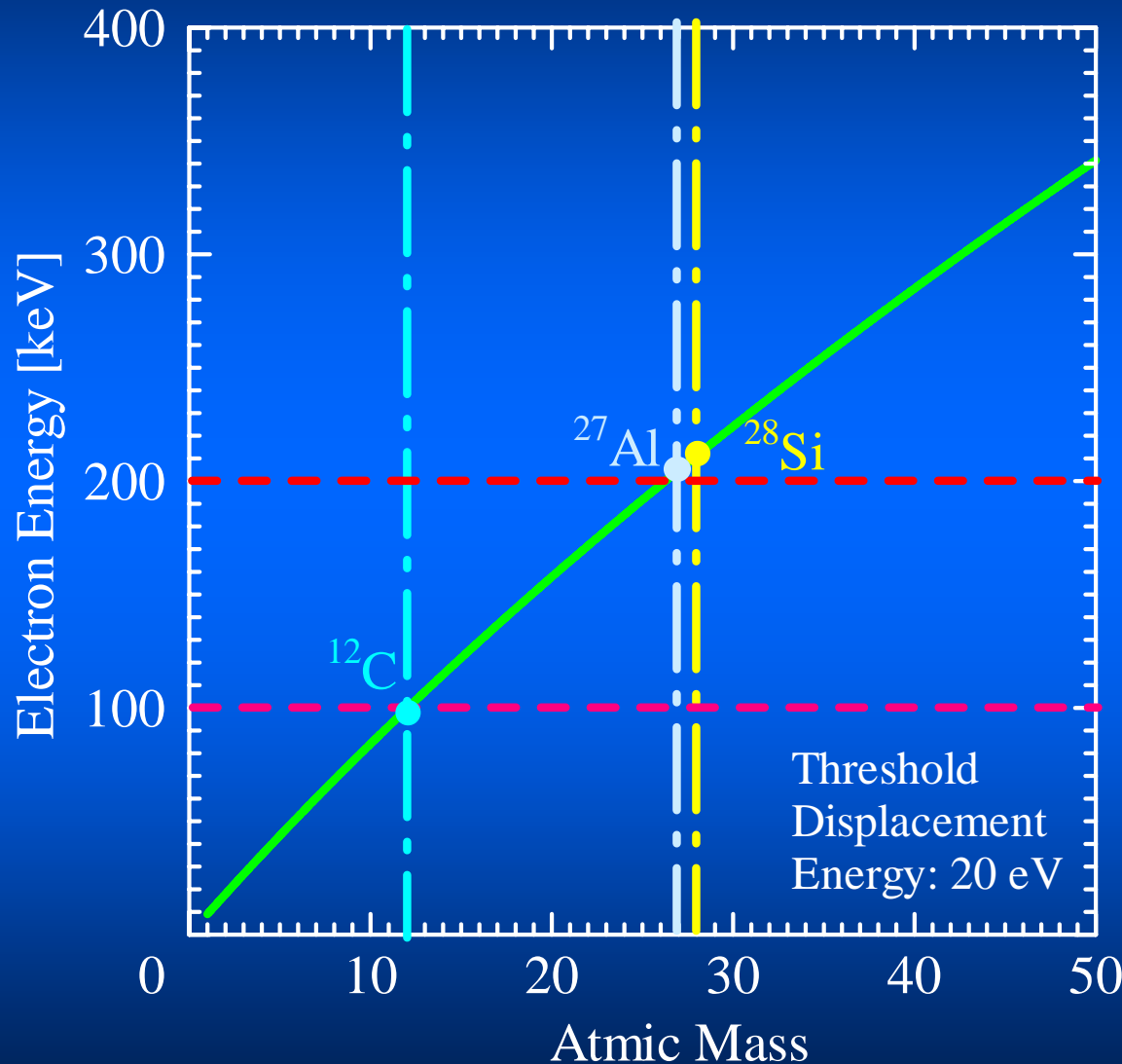


In the case of  $E_d=20$  eV, the substitutional C can be displaced by 100 keV electron irradiation.

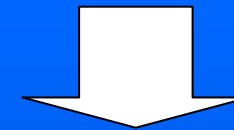


**However, substitutional Al and Si cannot be displaced.**

Minimum Electron Energy for displacement of Substitutional Atom



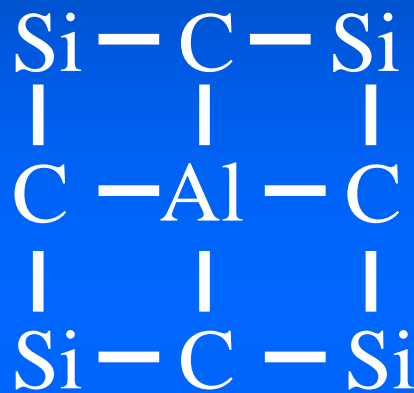
Experimentally, the 100 keV electron irradiation could not displace the substitutional C.



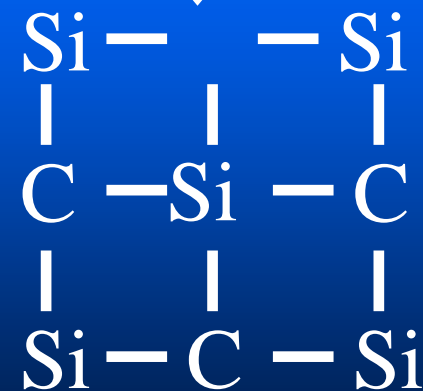
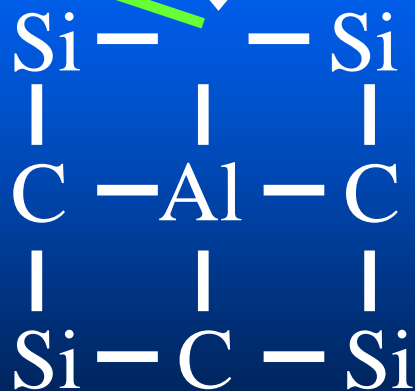
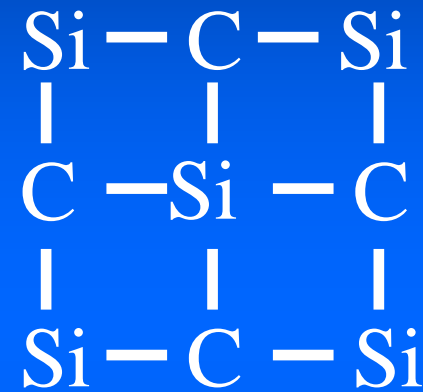
**The 200 keV electron irradiation cannot displace the substitutional Al and Si.**

# Displacement of C by 200 keV electron irradiation

Al acceptor



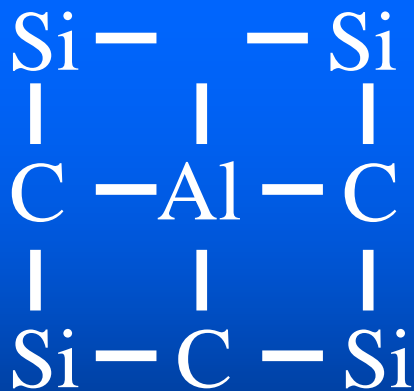
SiC matrix



# The reason why the $N_{Al}$ is reduced by irradiation

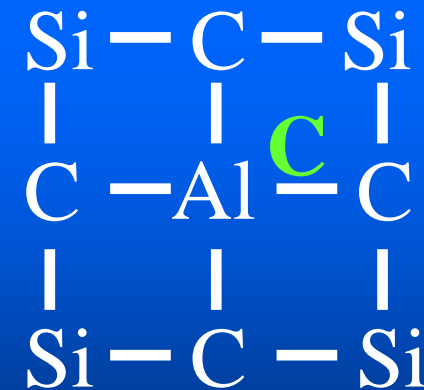


Displacement of  $C_s$



or

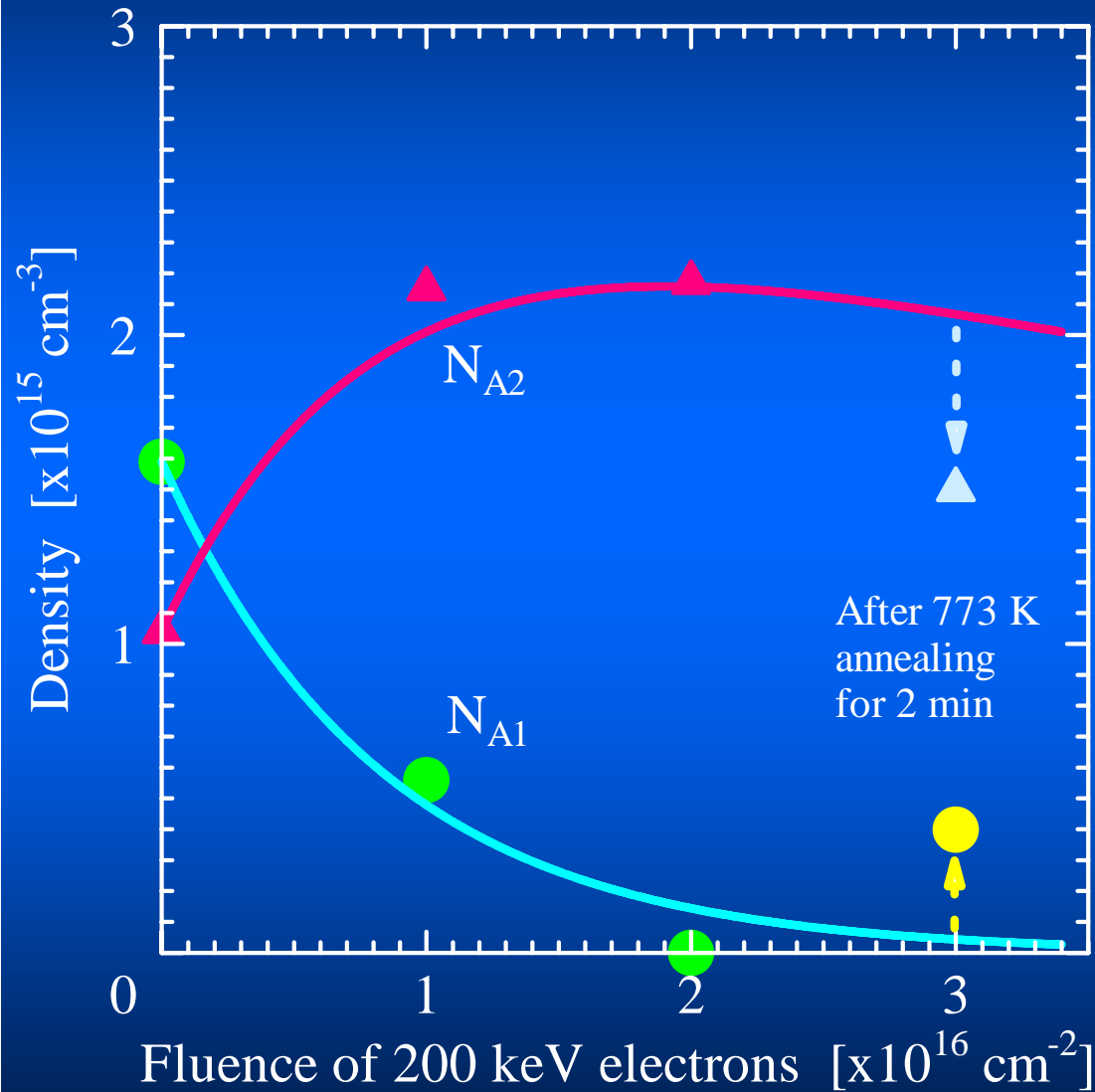
Migration of  $C_i$



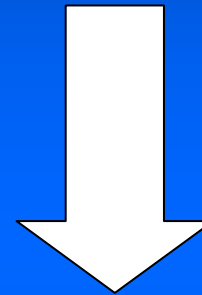
$Al_{Si}-V_C$  complex  $\rightarrow N_{DA} \leftarrow Al_{Si}-C_s-C_i$  complex

$N_{Al}$  decreases, whereas  $N_{DA}$  increases.

# Change of $N_{Al}$ and $N_{DA}$ in 6H-SiC by annealing

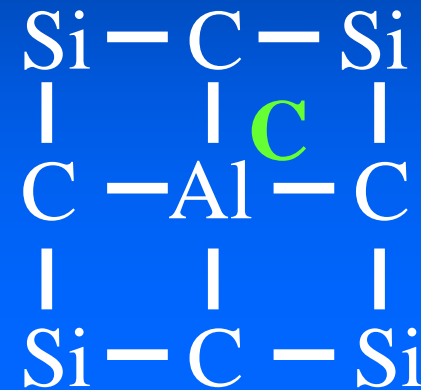
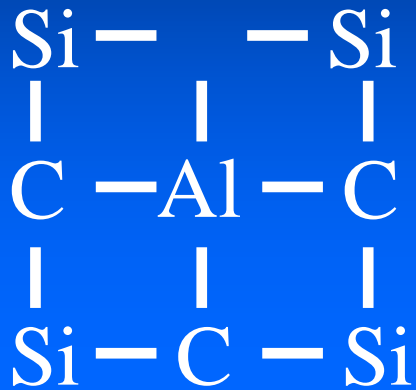


500 °C Annealing

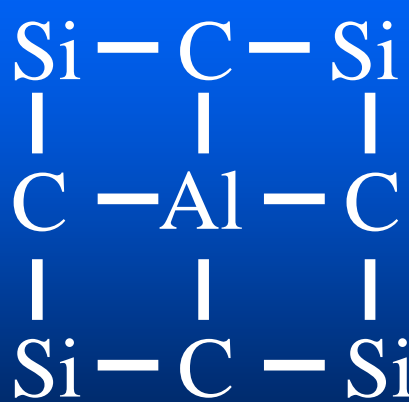


**$V_C$  cannot migrate,  
whereas  
 $C_i$  can migrate.**

The reason why the  $N_{Al}$  is increased by annealing



Migration of  $C_i$



Al acceptor

$N_{Al}$

$N_{Al}$  increases, whereas  $N_{DA}$  decreases.



## Conclusion

1. With 200 keV electron irradiation, the Al acceptor density ( $N_{Al}$ ) was decreased, while the unknown deep acceptor density ( $N_{DA}$ ) was increased.
2. 200 keV electron could displace only a substitutional C into an interstitial site.
3. With 500 °C annealing, the increment of  $N_{Al}$  was similar to the decrement of  $N_{DA}$ .
4. The deep acceptor might be an  $Al_{Si}-V_C$  complex or  $Al_{Si}-C_s-C_i$  complex.